

# Uncertainty quantification for solute transport modeling

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**Motivation:** who do we blame when experiments and theory don't agree?

... or: how can we agree if we don't know what we're agreeing to?



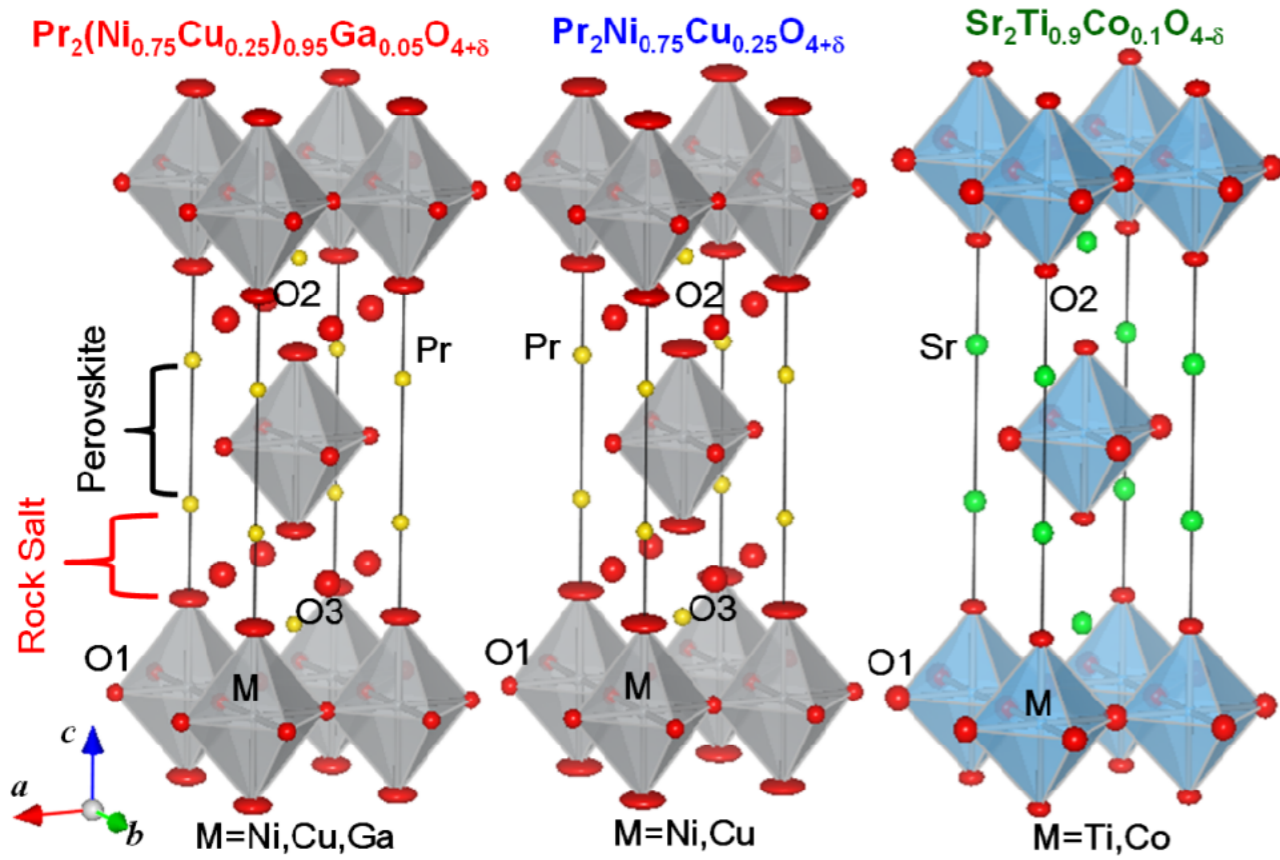
Propagating uncertainty from first-principles calculations via analytic models of mass transport can quantify uncertainty in diffusivity predictions.



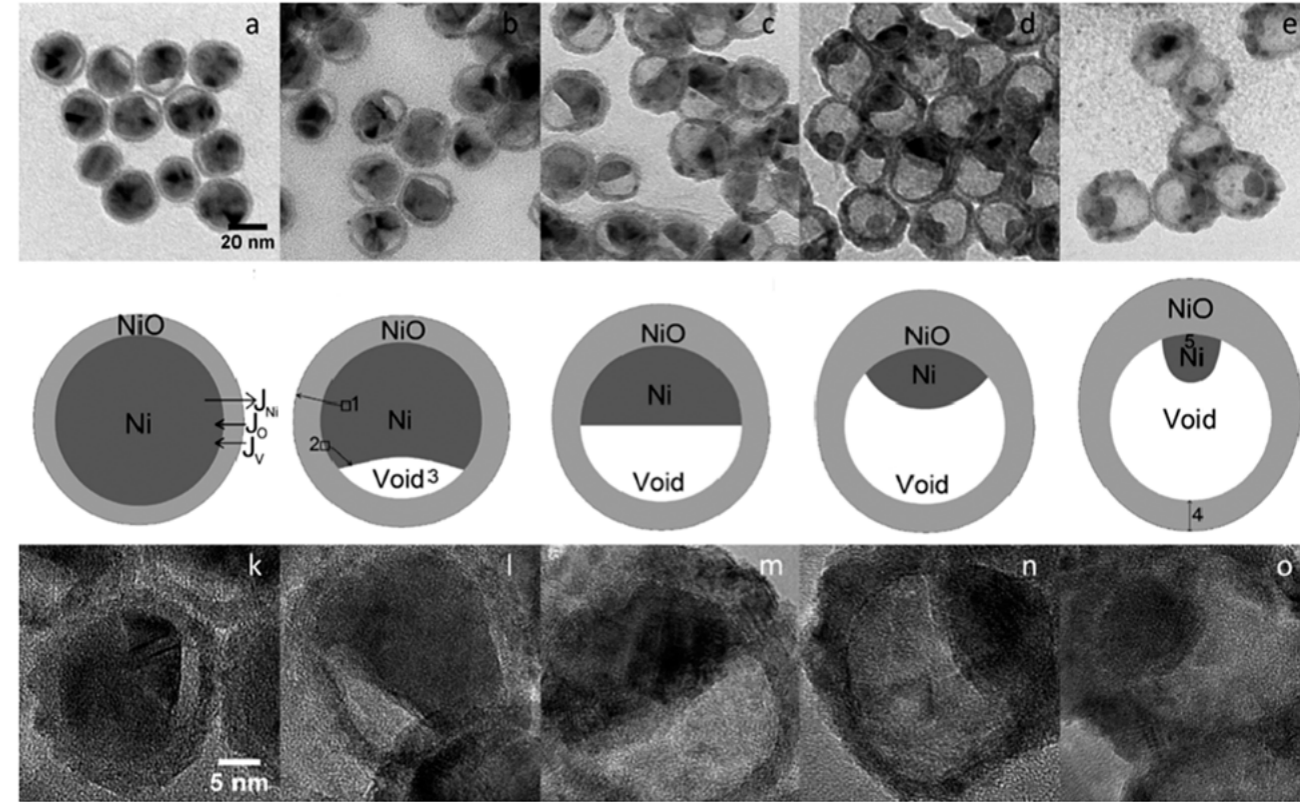
[dtrinkle.matse@illinois.edu](mailto:dtrinkle.matse@illinois.edu)

# Mass transport

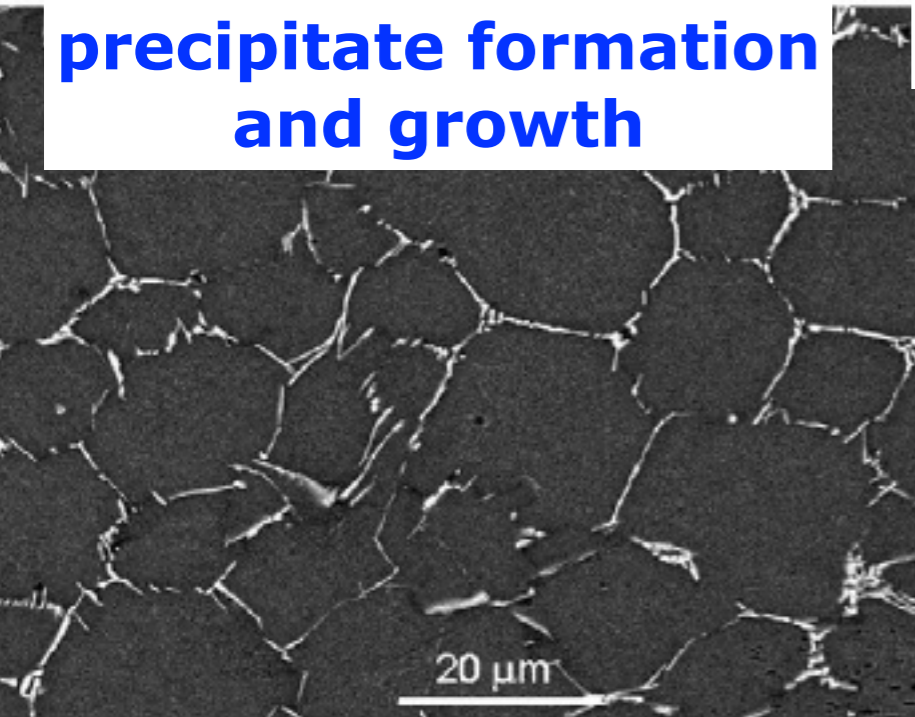
## ion transport



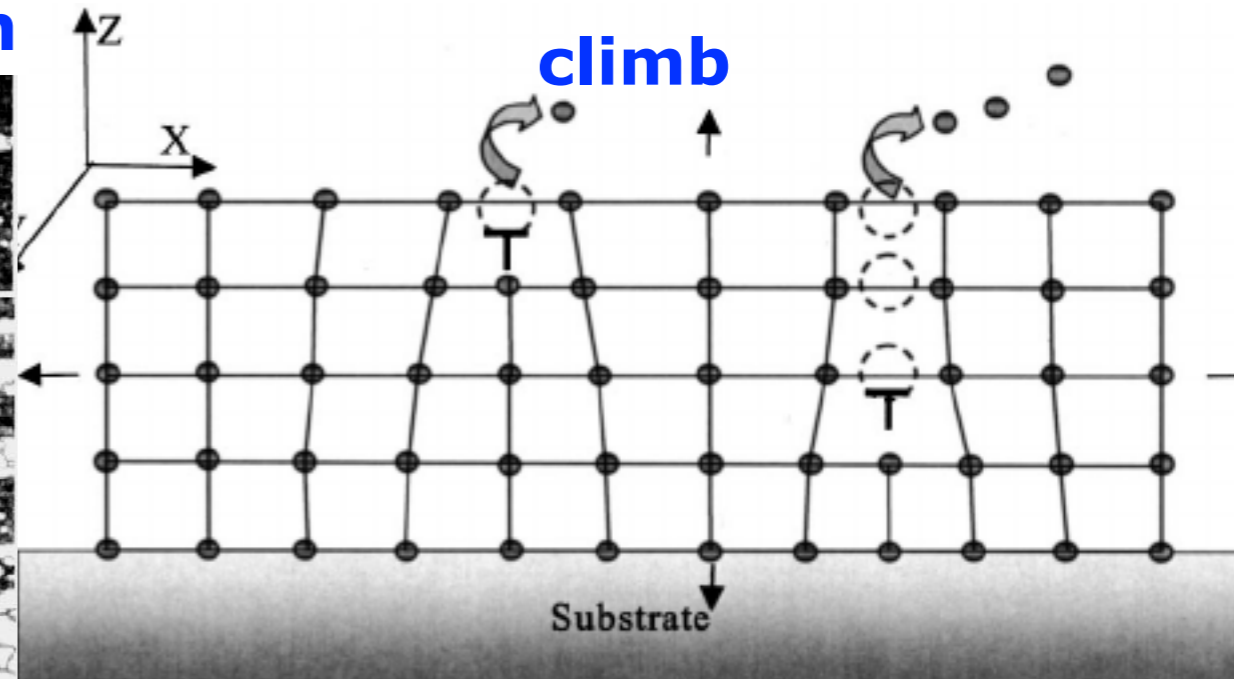
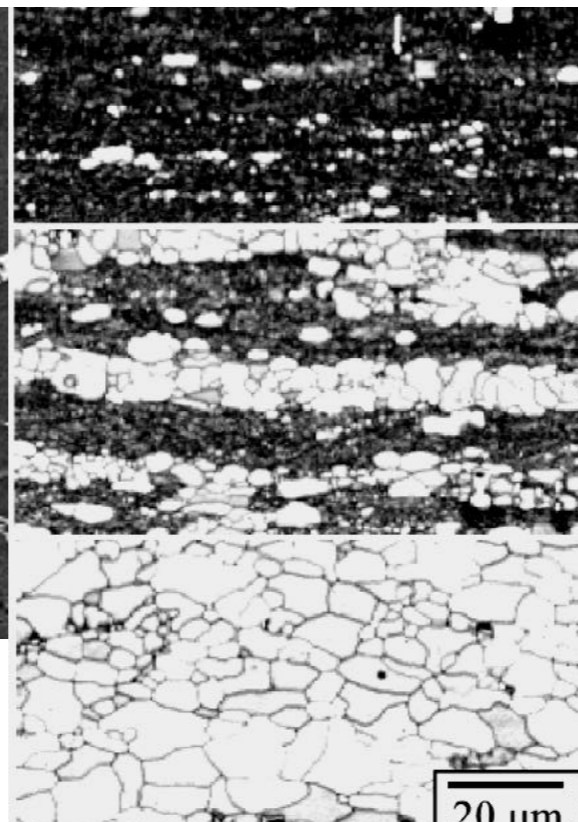
## (nano) Kirkendall effect



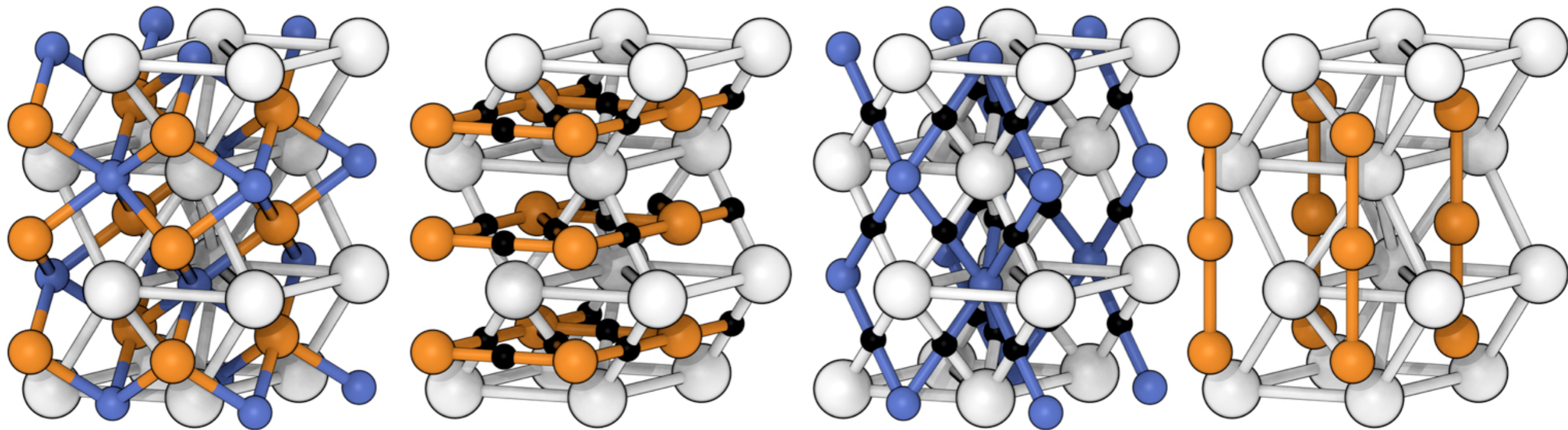
## precipitate formation and growth



## recrystallization



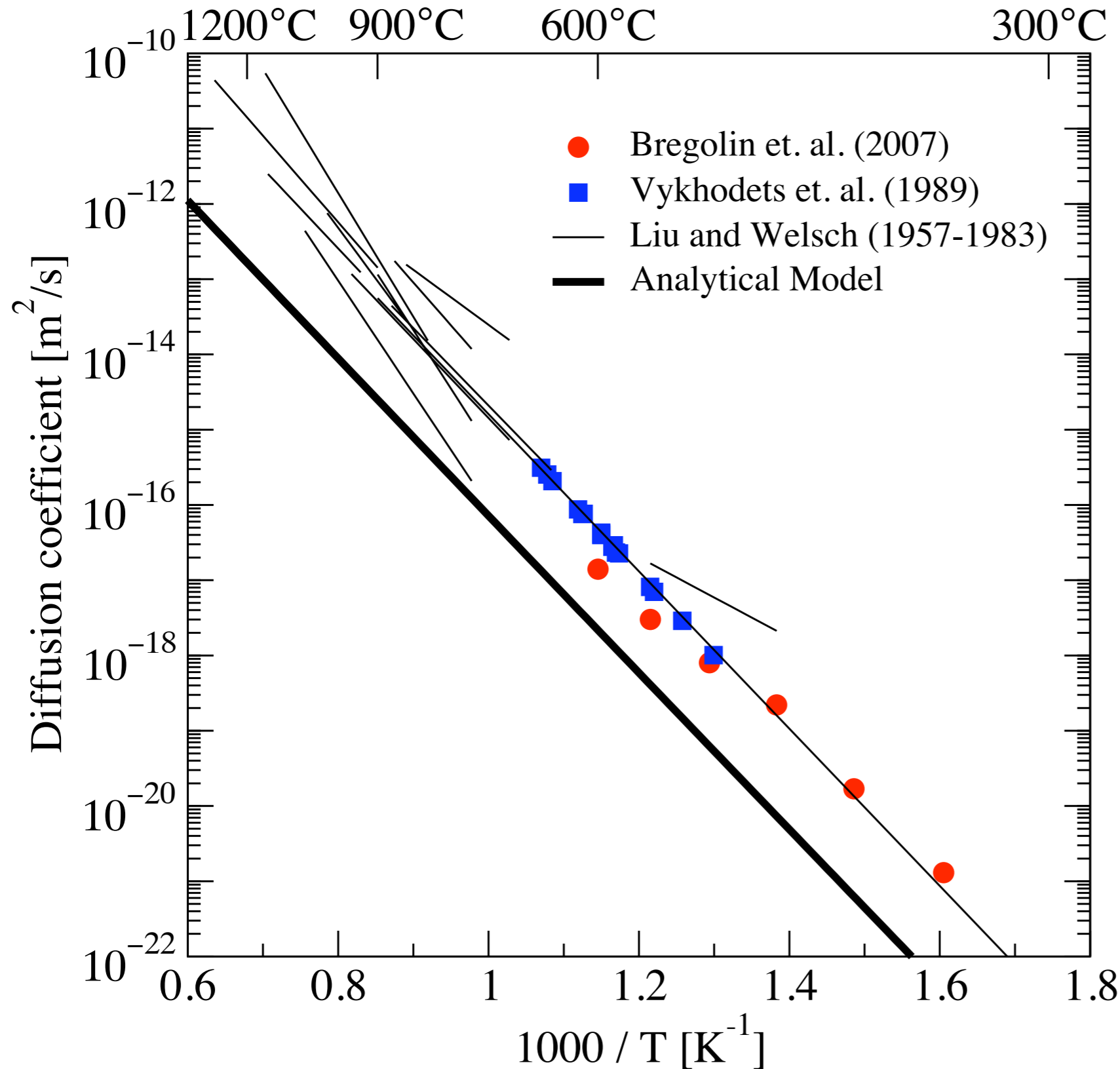
# Oxygen diffusion constant in HCP Ti



$$\begin{aligned}
 D_{\text{basal}} &= a_{\text{Ti}}^2 \left[ \begin{array}{c} \lambda_{\text{oh}} \\ \frac{3}{8}\lambda_{\text{oh}} \end{array} + \begin{array}{c} \frac{3}{4}\lambda_{\text{oc}} \\ 0\lambda_{\text{oc}} \end{array} + \begin{array}{c} \frac{1}{4}\lambda_{\text{hc}} \begin{pmatrix} \lambda_{\text{oh}} \\ \lambda_{\text{ho}} \end{pmatrix} \\ \frac{3}{8}\lambda_{\text{hc}} \begin{pmatrix} \lambda_{\text{oh}} \\ \lambda_{\text{ho}} \end{pmatrix} \end{array} + \begin{array}{c} 0\lambda_{\text{oo}} \\ \frac{1}{4}\lambda_{\text{oo}} \end{array} \right] \\
 D_{\text{c-axis}} &= c_{\text{Ti}}^2 \left[ \begin{array}{c} \lambda_{\text{oh}} \\ \frac{3}{8}\lambda_{\text{oh}} \end{array} + \begin{array}{c} \frac{3}{4}\lambda_{\text{oc}} \\ 0\lambda_{\text{oc}} \end{array} + \begin{array}{c} \frac{1}{4}\lambda_{\text{hc}} \begin{pmatrix} \lambda_{\text{oh}} \\ \lambda_{\text{ho}} \end{pmatrix} \\ \frac{3}{8}\lambda_{\text{hc}} \begin{pmatrix} \lambda_{\text{oh}} \\ \lambda_{\text{ho}} \end{pmatrix} \end{array} + \begin{array}{c} 0\lambda_{\text{oo}} \\ \frac{1}{4}\lambda_{\text{oo}} \end{array} \right]
 \end{aligned}$$

Oxygen diffusion separates into the sum of the single networks.

# Oxygen diffusion constant in HCP Ti



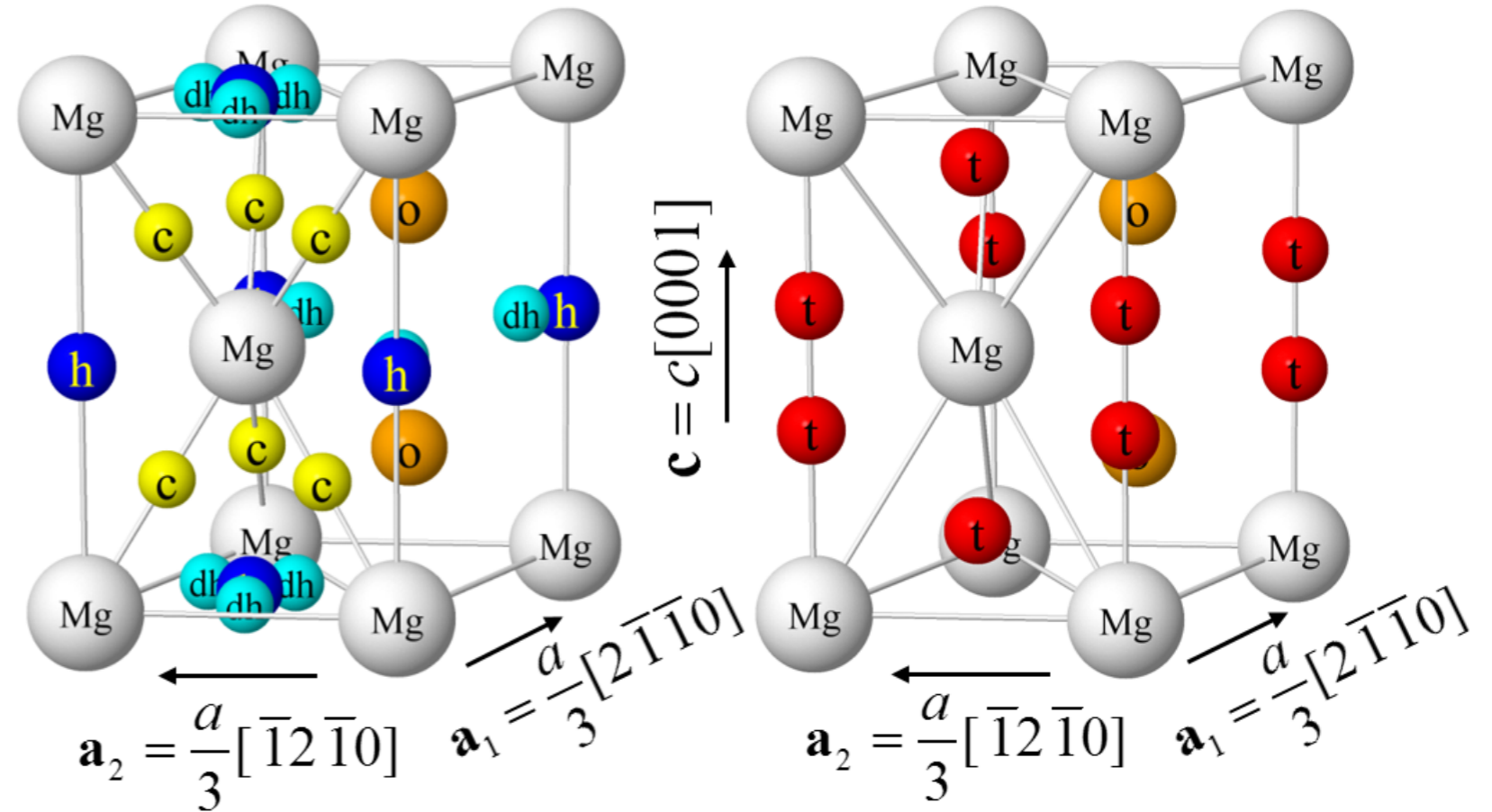
Analytical results a factor of 10 lower than experimental observations.

Analytical barriers match well to experiments.

$$D_0 = 2.18 \times 10^{-6} \text{ m}^2\text{s}^{-1}$$
$$E_a = 2.08 \text{ eV}$$

# Interstitial sites in HCP Mg

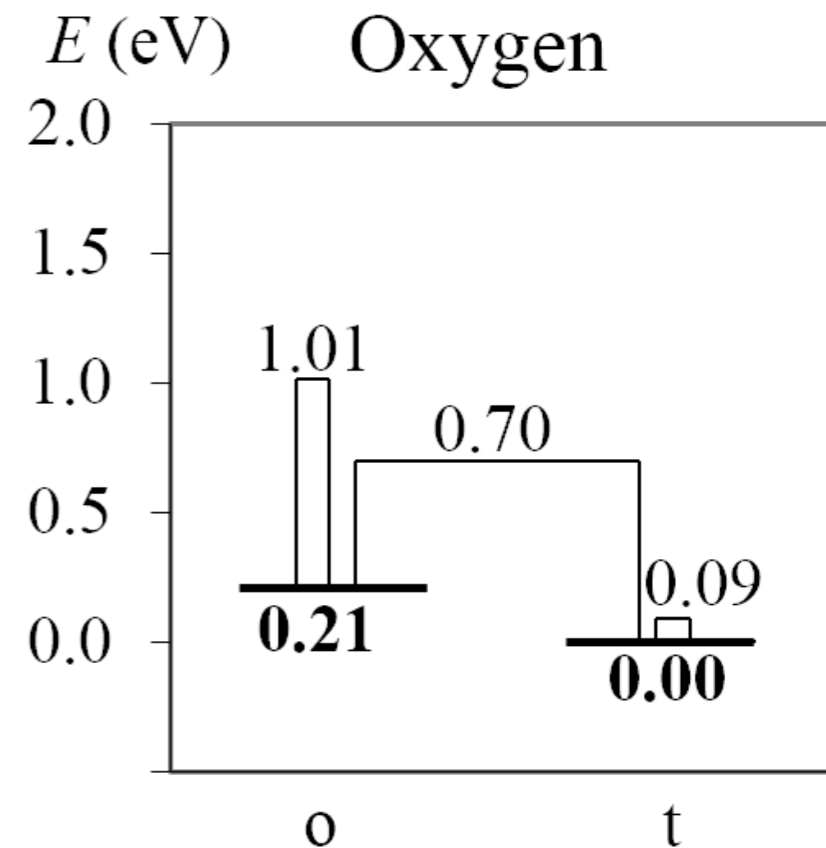
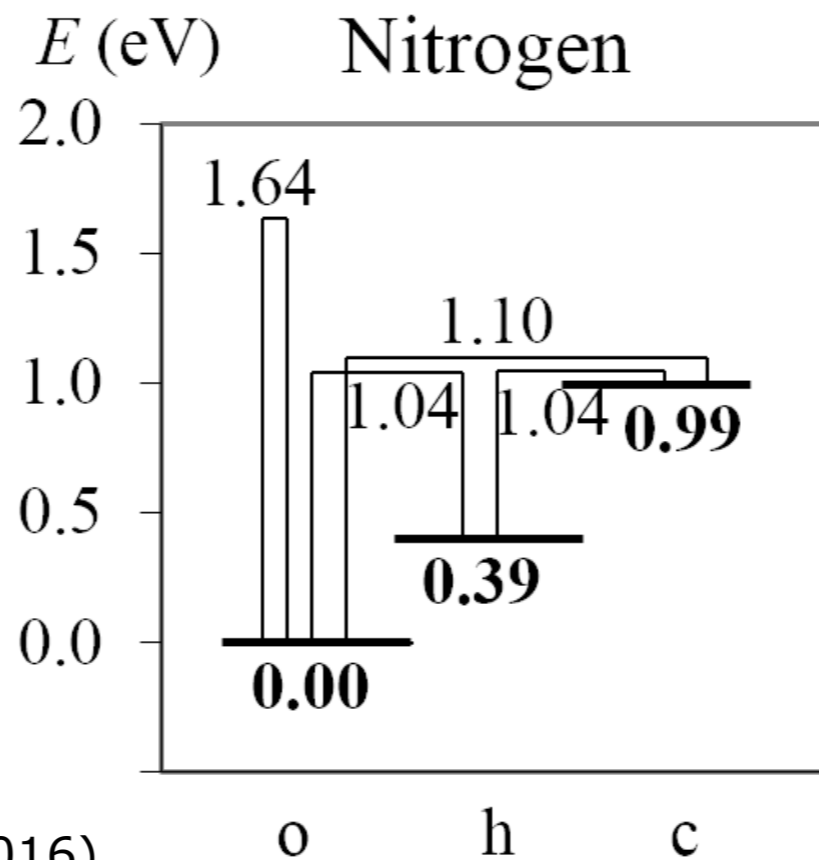
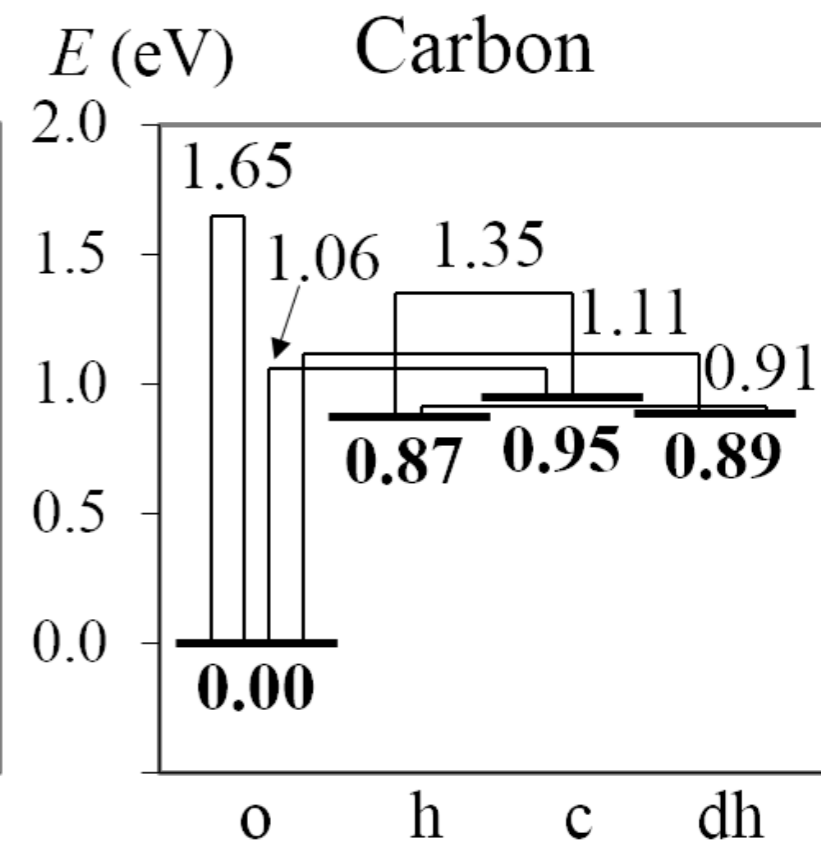
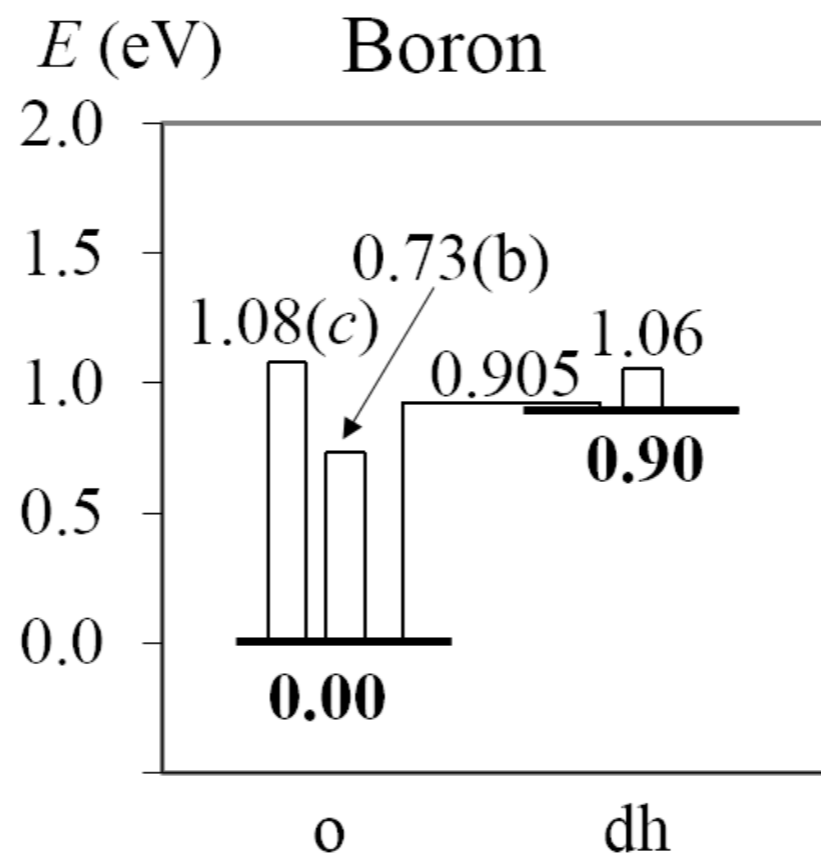
- Tetrahedral (t)
- Octahedral (o)
- Hexahedral (h)
- Crowdion (c)
- Distorted Hexahedral (dh)



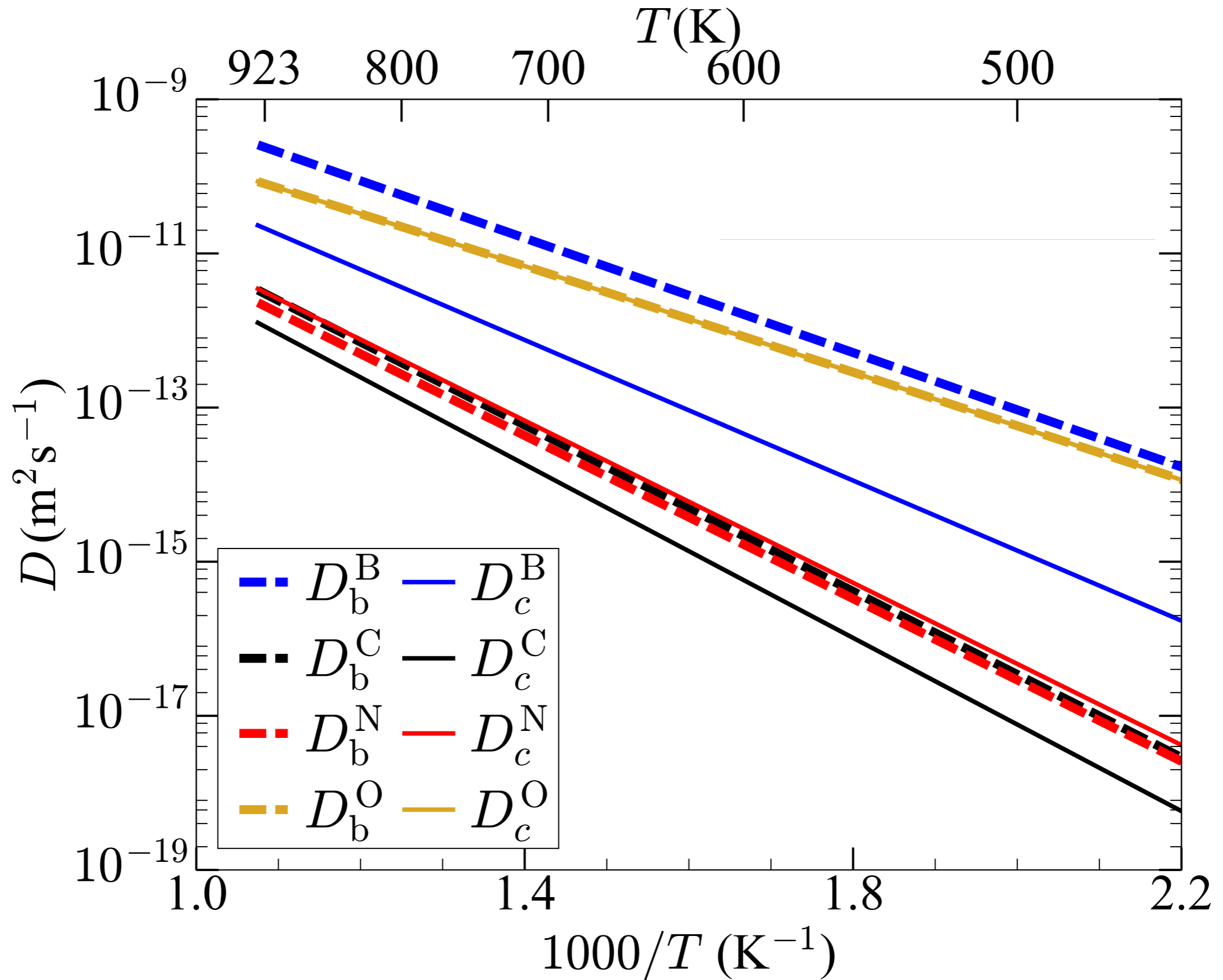
Solute	Ground state	Metastable (energy in eV over ground state) $3s^2$	Metastable (energy in eV over ground state) $3s^2 2p^6$
Boron	Octahedral	dh (0.90)	dh (0.89)
Carbon	Octahedral	h (0.87), dh (0.89), c (0.94)	h (0.87), dh (0.88), c (0.94)
Nitrogen	Octahedral	h (0.39), c (0.99)	h (0.39), c (0.99)
Oxygen	Tetrahedral	o (0.21)	o (0.19)

# Migration barriers in HCP Mg

- Rate of migration  $\sim e^{-E_b/kBT}$
- Migration barrier from CI-NEB with one intermediate image



# Diffusivity in HCP Mg



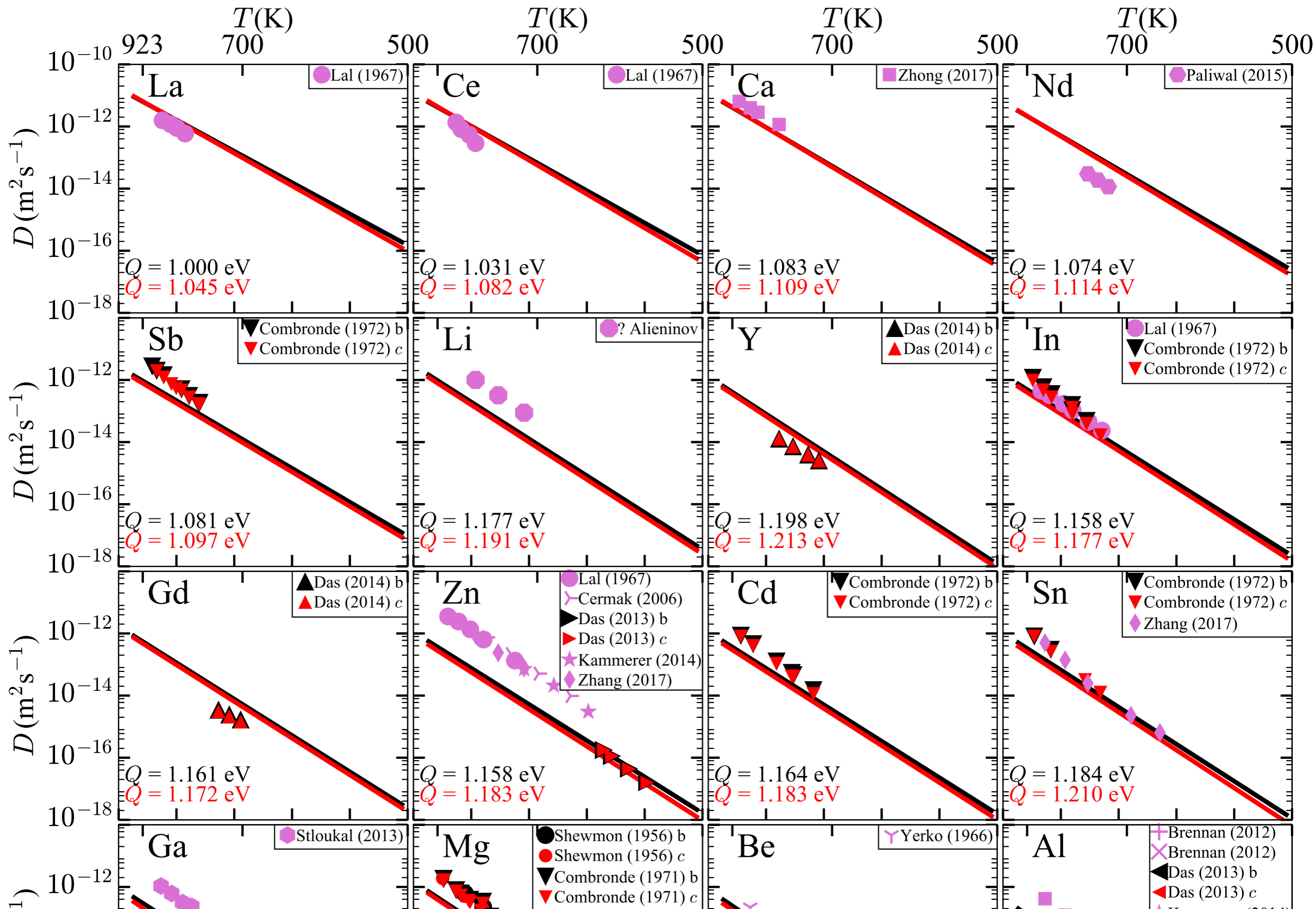
# Vacancy mediated diffusion in Mg

- Differences of  $\sim 0.1$  eV in activation energies for rare-earth solutes
- Significant differences in crossover temperatures

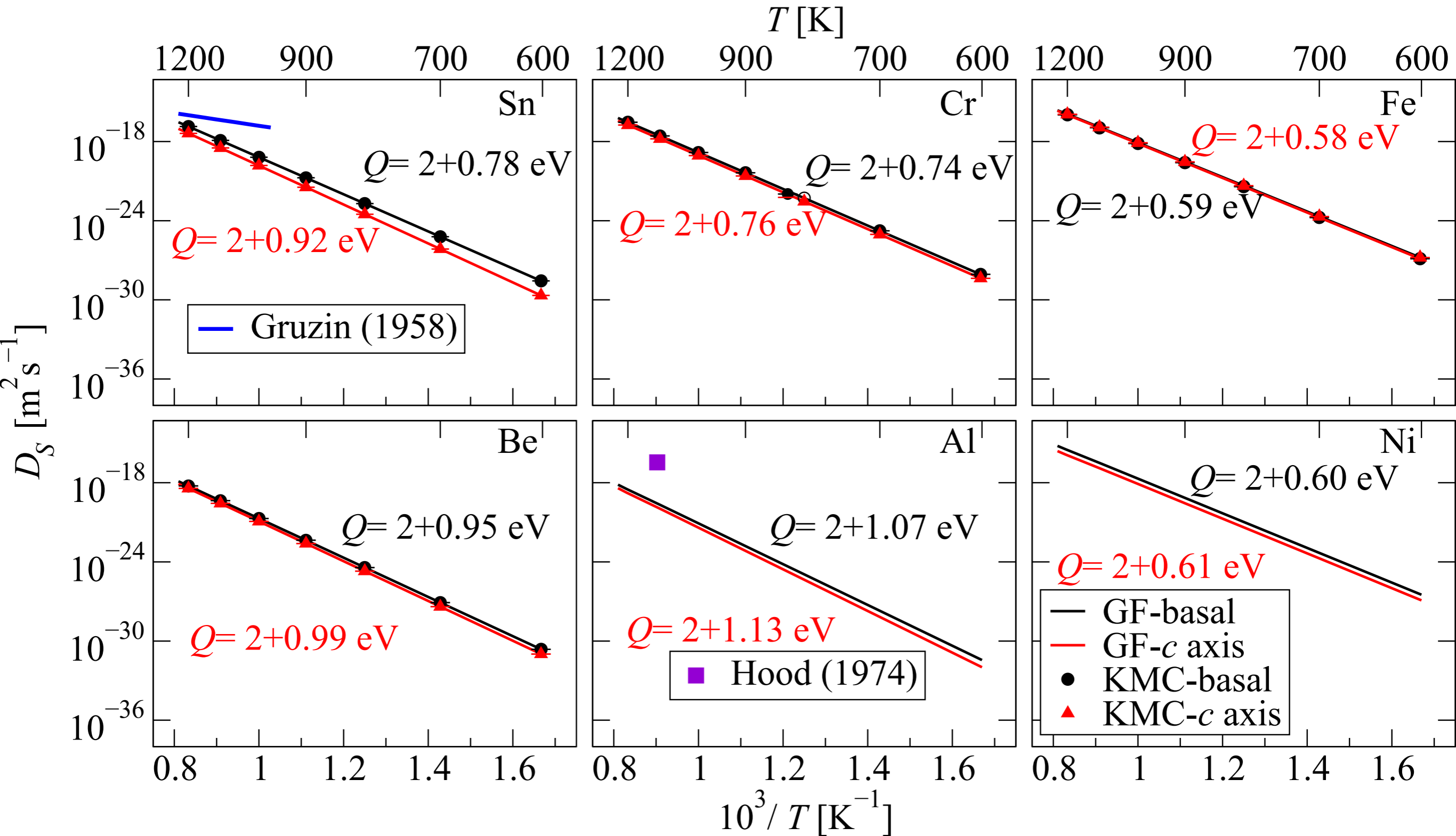
Solute	8-freq. ( $Q$ )	GF ( $Q$ )	Exp. ( $Q$ )	GF ( $T_{\text{cross}}$ )	8-f ( $T_{\text{cross}}$ )
Nd	1.18   1.20	1.08   1.13	1.16 (Paliwal2015)	529   422	< 0 K
Ce	1.14   1.15	1.03   1.09	1.82 (Lal1966)	648   589	252   287
La	1.10   1.11	1.00   1.04	1.06 (Lal1966)	746   714	415   452
Gd	1.24   1.26	1.16   1.17	0.82   0.85 (Das2014)	341   218	> 923 K
Y	1.25   1.27	1.20   1.21	1.01   1.02 (Das2014)	271   190	> 923 K
Ca	1.12   1.14	1.08   1.11	1.07 (Zhong2017)	538   501	309   343



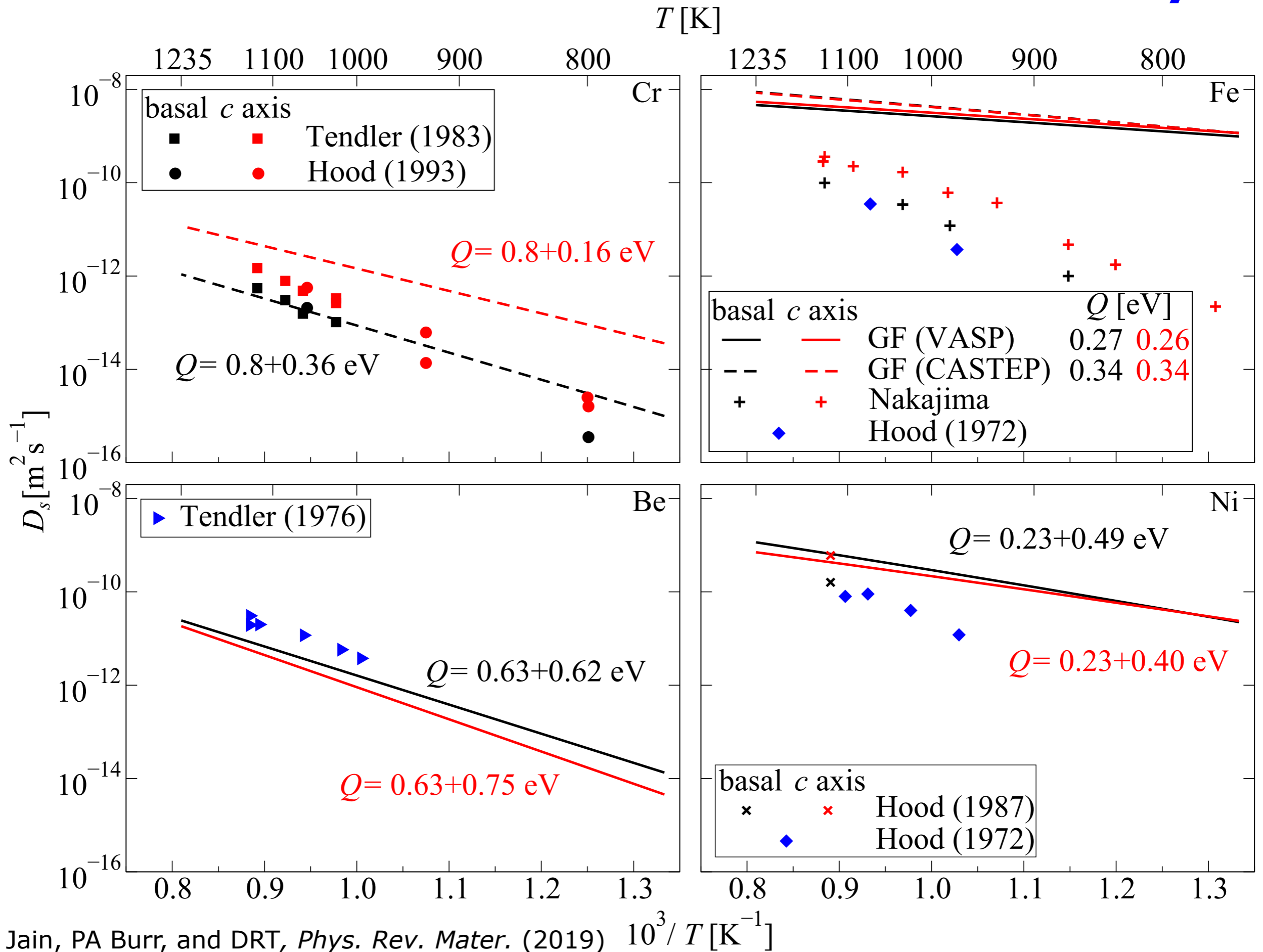
# Vacancy mediated diffusion in Mg



# Vacancy mediated diffusion in Zr: solute diffusivity



# Interstitial diffusion in Zr: solute diffusivity

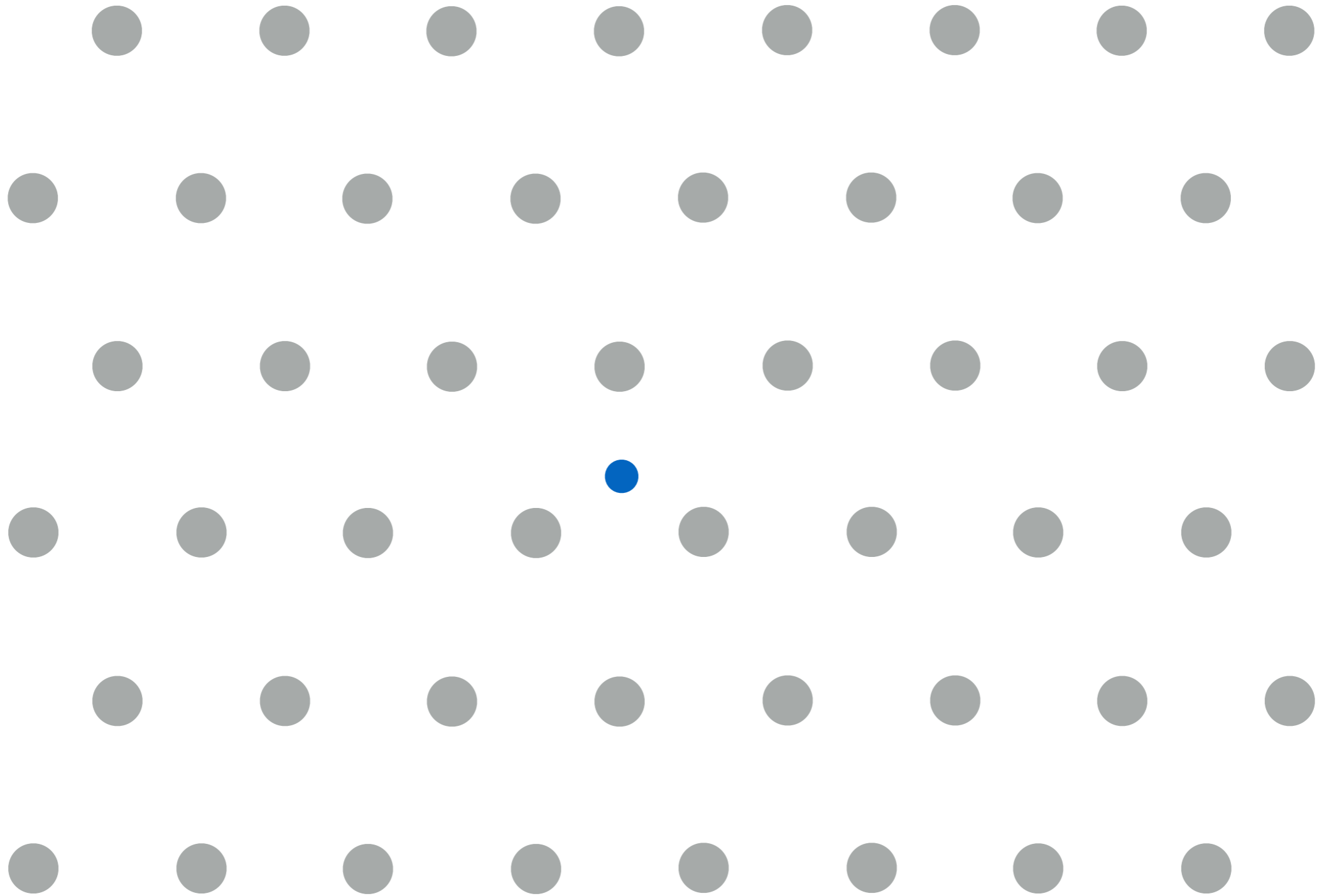


# What's the disagreement all about?

- *Are there experimental issues?*
  - Adequate thermal control?
  - Are point defects at equilibrium?
  - Is the species transport compositionally insensitive?
- *Are there computational issues?*
  - Do we have the proper diffusion mechanism?
  - Is DFT incorrect?
- *Are our expectations wrong?*
  - What sort of agreement *should* we be expecting?

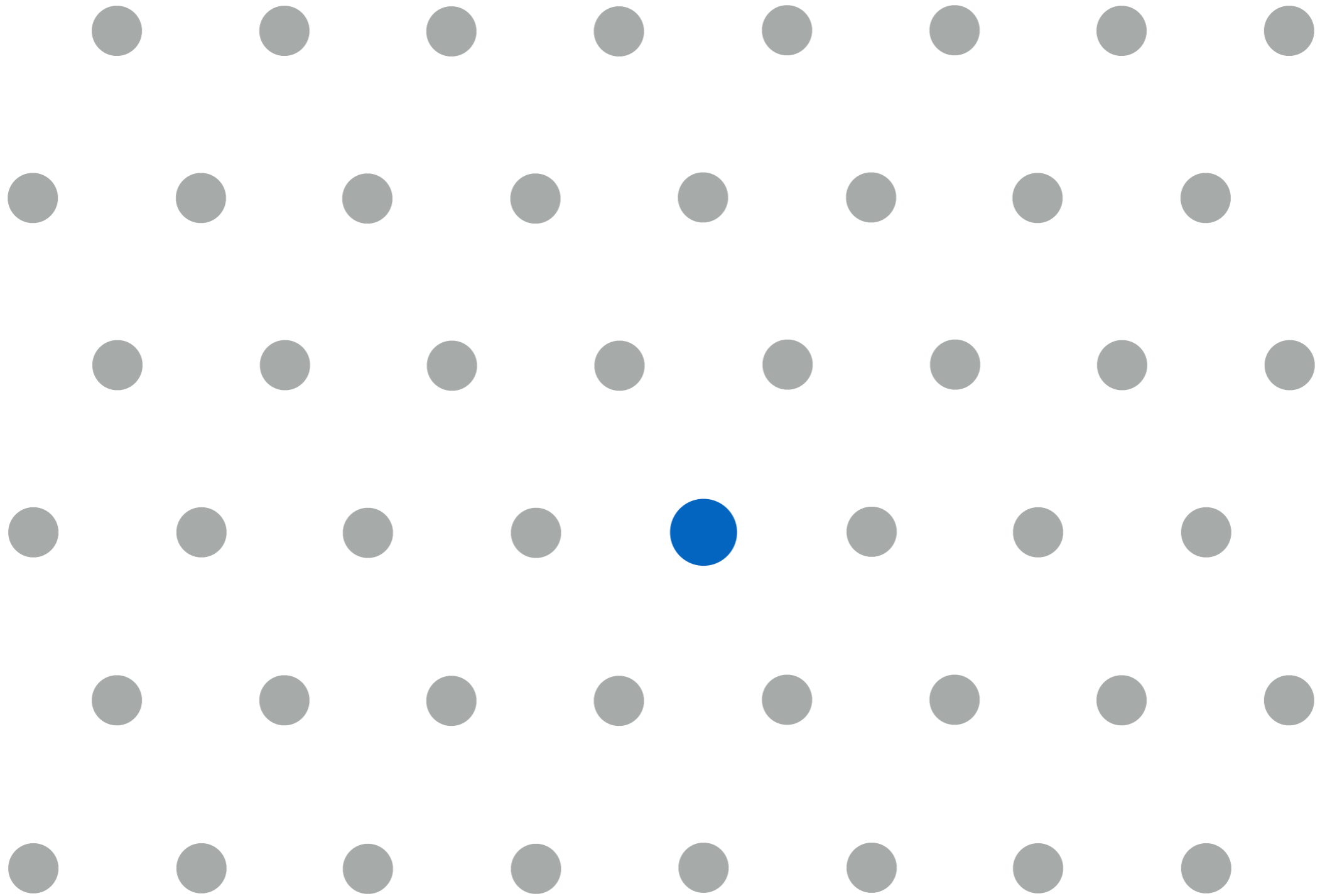
Can we (1) quantify DFT uncertainty and (2) propagate that forward to quantify transport modeling uncertainty?

# Atomistic mechanism for diffusivity: interstitial solute



$$\lim_{t \rightarrow \infty} \left\langle \frac{(x_i(t) - x_i(0))(x_j(t) - x_j(0))}{2t} \right\rangle = D_{ij}$$

# Atomistic mechanism for diffusivity: substitutional solute



$$\lim_{t \rightarrow \infty} \left\langle \frac{(x_i^A(t) - x_i^A(0))(x_j^B(t) - x_j^B(0))}{2t \Omega k_B T} \right\rangle = L_{ij}^{AB}$$

# Replacing trajectories with probabilities: Master equation

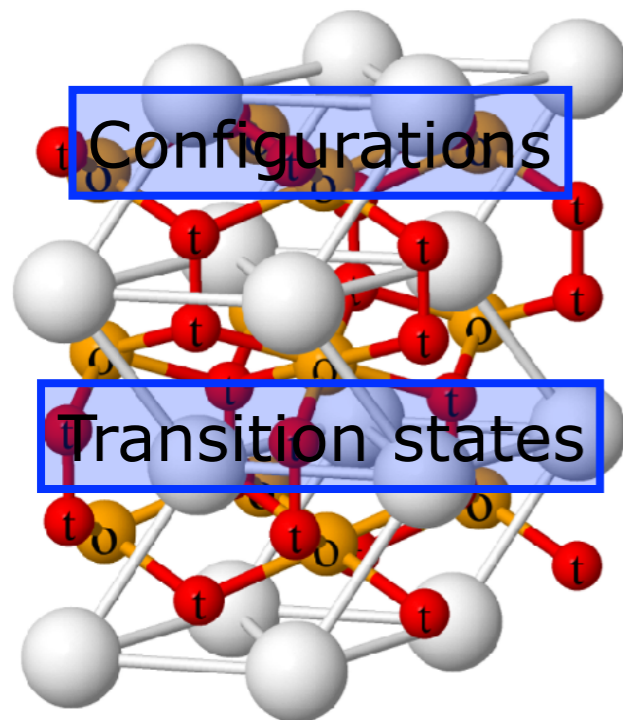
$$\frac{dP(\underline{\chi}, t)}{dt} = \sum_{\underline{\chi}'} P(\underline{\chi}', t) W(\underline{\chi}' \rightarrow \underline{\chi}) - P(\underline{\chi}, t) W(\underline{\chi} \rightarrow \underline{\chi}')$$
$$= \sum_{\underline{\chi}'} P(\underline{\chi}', t) W_{\underline{\chi}'\underline{\chi}}$$

configuration  $\underline{\chi}$ , probability  $P(\underline{\chi}, t)$

## • Master equation:

- Well-defined states that thermalize before next transition
- Markovian process (no memory)
- Equilibrium: detailed balance (no fluxes)
- Steady state: balance (constant fluxes, no time evolution)

### Geometry



### Computational evaluation

*DFT*: supercell  
relaxation + NEB

### Data output

Configuration  
energies + entropies

Transition  
energies + entropies

Software:

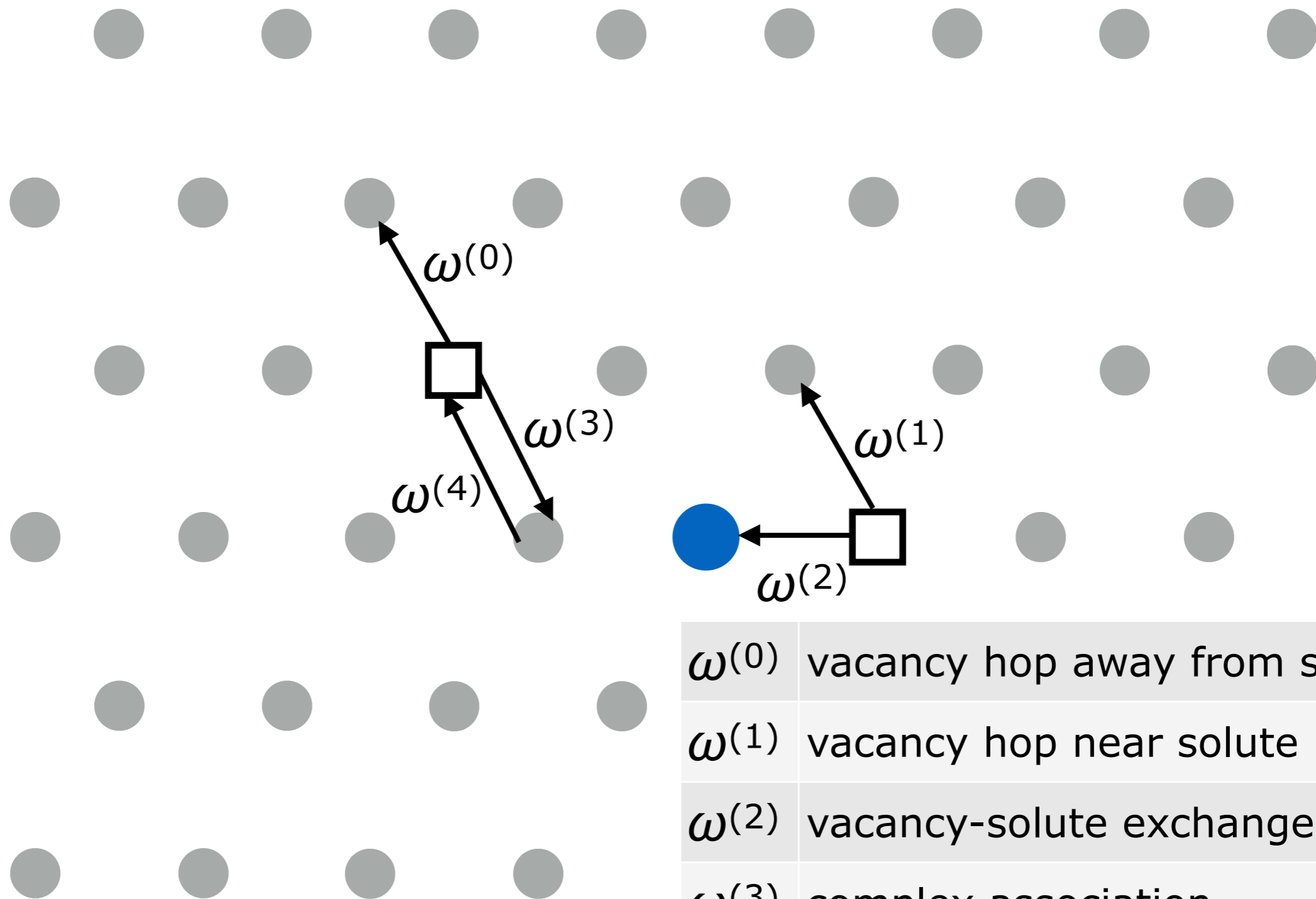
[dallastrinkle.github.io/Onsager](https://github.com/dallastrinkle/Onsager)

Interstitials:

Trinkle, *Phil. Mag.* **96** (2016)

Vacancy-mediated: Trinkle, *Phil. Mag.* **97** (2017)

# Atomistic mechanism for diffusivity: "five frequency model"



$\omega^{(0)}$	vacancy hop away from solute
$\omega^{(1)}$	vacancy hop near solute
$\omega^{(2)}$	vacancy-solute exchange
$\omega^{(3)}$	complex association
$\omega^{(4)}$	complex dissociation



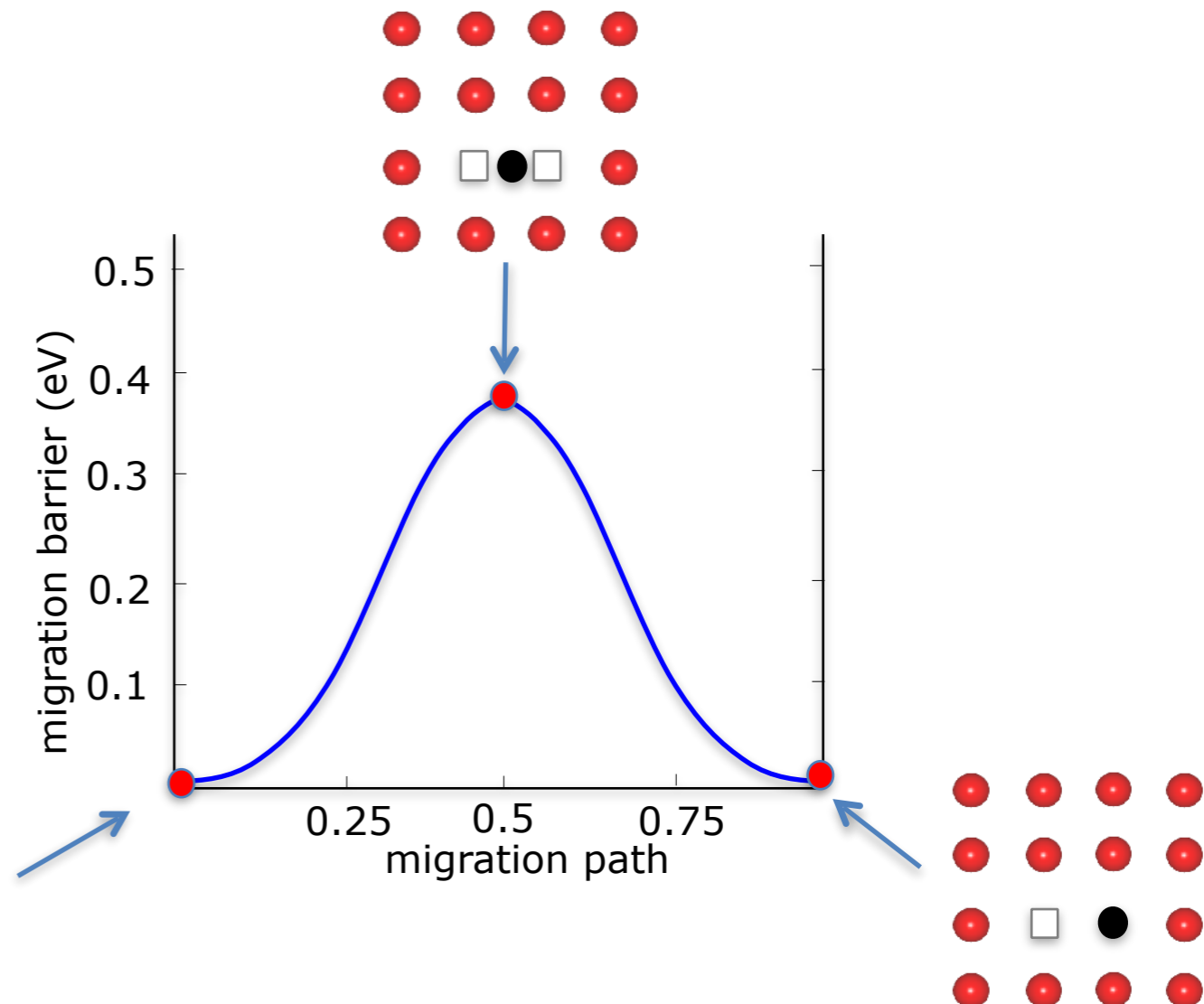
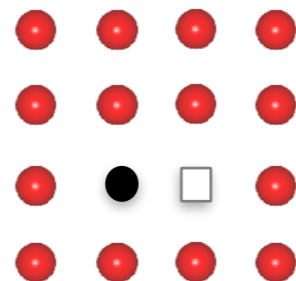
# Jump frequencies: harmonic transition state theory

atomic jump rate: 
$$\omega = \frac{\prod_{3N} \nu^{(\text{initial state})}}{\prod_{3N-1} \nu^{(\text{final state})}} \exp\left(-\frac{\Delta F_{\text{elec}}^{(\text{TS-IS})}}{k_B T}\right) \exp\left(-\frac{\Delta H_{\text{mig}}^{(\text{TS-IS})}}{k_B T}\right)$$

phonon frequency

enthalpy of migration

- First-principles (DFT) calculation of:
  - transition state using nudged-elastic band
  - phonon frequencies from finite difference force constants
  - electronic contribution from density of states (ignorable)



# Computing diffusivity: random walk

$$\mathbf{x}(t) - \mathbf{x}(0) = \sum_n \delta \mathbf{x}_n \quad \text{Atomic jumps}$$

$$(\mathbf{x}(t) - \mathbf{x}(0)) \otimes (\mathbf{x}(t) - \mathbf{x}(0)) = \sum_{nn'} \delta \mathbf{x}_n \otimes \delta \mathbf{x}_{n'}$$

$$= \sum_n \left\{ \delta \mathbf{x}_n \otimes \delta \mathbf{x}_n + 2 \sum_{m=1}^{\infty} \delta \mathbf{x}_n \otimes \delta \mathbf{x}_{n+m} \right\}$$

“bare” mean-squared displacement

correlation

- Stochastic methods:
  - Molecular dynamics (including accelerated approaches)
  - Kinetic Monte Carlo<sup>1</sup> (including on-the-fly)
- *Multiple issues:*
  - **Stochastic:** increasing number of trajectories to reduce variance
  - **Correlation:**
    - increasingly long trajectories to reduce error
    - increasingly size cells to reduce error
    - poor convergence for systems with large rate anisotropy
  - Evaluating derivatives with respect to strain converges poorly<sup>2</sup>

<sup>1</sup>G. E. Murch, *Diffusion in Crystalline Solids*, (1984) Chap. 7

<sup>2</sup>Li and Trinkle, *Phys. Rev. E* **96** (2016)

# Computing diffusivity: Direct solution of master equation

$$\frac{dP(\underline{\chi}, t)}{dt} = \sum_{\underline{\chi}'} P(\underline{\chi}', t) W(\underline{\chi}' \rightarrow \underline{\chi}) - P(\underline{\chi}, t) W(\underline{\chi} \rightarrow \underline{\chi}')$$

$$= \sum_{\underline{\chi}'} P(\underline{\chi}', t) W_{\underline{\chi}'\underline{\chi}} \quad \text{configuration } \underline{\chi}, \text{ probability } P(\underline{\chi}, t)$$

$$P_0(\underline{\chi}) W(\underline{\chi} \rightarrow \underline{\chi}') = P_0(\underline{\chi}') W(\underline{\chi}' \rightarrow \underline{\chi}) \quad \text{equilibrium (no flux)}$$

$$\sum_{\underline{\chi}'} P_{ss}(\underline{\chi}) W(\underline{\chi} \rightarrow \underline{\chi}') = \sum_{\underline{\chi}'} P_{ss}(\underline{\chi}') W(\underline{\chi}' \rightarrow \underline{\chi}) \quad \text{steady state (constant flux)}$$

$$\mathbf{J}^\alpha = \frac{1}{2\Omega} \sum_{\underline{\chi}\underline{\chi}'} \left\{ P_{ss}(\underline{\chi}) W_{\underline{\chi}\underline{\chi}'} - P_{ss}(\underline{\chi}') W_{\underline{\chi}'\underline{\chi}} \right\} \delta \mathbf{x}^\alpha(\underline{\chi} \rightarrow \underline{\chi}') = - \sum_{\beta} \underline{\mathbf{L}}^{\alpha\beta} \nabla \mu_\beta$$

$$\mathbf{b}^\alpha(\underline{\chi}) := \sum_{\underline{\chi}'} W_{\underline{\chi}\underline{\chi}'} \delta \mathbf{x}^\alpha(\underline{\chi} \rightarrow \underline{\chi}') \quad g_{\underline{\chi}\underline{\chi}'} := W_{\underline{\chi}\underline{\chi}'}^+ \text{ (pseudoinverse)}$$

$$\underline{\mathbf{L}}^{\alpha\beta} = \frac{1}{k_B T \Omega} \sum_{\underline{\chi}\underline{\chi}'} \frac{1}{2} P_0(\underline{\chi}) W_{\underline{\chi}\underline{\chi}'} \delta \mathbf{x}^\alpha(\underline{\chi} \rightarrow \underline{\chi}') \otimes \delta \mathbf{x}^\beta(\underline{\chi} \rightarrow \underline{\chi}') \quad \text{"bare" mean-squared displacement}$$

$$+ \frac{1}{k_B T \Omega} \sum_{\underline{\chi}\underline{\chi}'} P_0(\underline{\chi}) \mathbf{b}^\alpha(\underline{\chi}) g_{\underline{\chi}\underline{\chi}'} \mathbf{b}^\beta(\underline{\chi}')$$

correlation

Interstitials: Trinkle, Phil. Mag. **96** (2016)  
 Vacancy-mediated: Trinkle, Phil. Mag. **97** (2017)

# Quantifying uncertainty in DFT

- **Density functional theory is not exact**
  - Controlled approximations: basis size, supercell size,  $k$ -point integration
  - Uncontrolled approximations: exchange-correlation potential
- **No more than one exchange-correlation treatment is correct**
  - Local-density approximation (LDA): Ceperley-Alder
  - Generalized gradient approximation (GGA): PBE
  - Meta-GGA: SCAN
  - Hybrid functionals
  - ... and many, many more
- **How can we quantifying uncertainty in DFT?**
  - Bayesian approach: sample over different exchange-correlation treatments, weighted by likelihood (e.g., *PRL* **95**, 216401 (2005))
  - Empirical estimate from variability in results

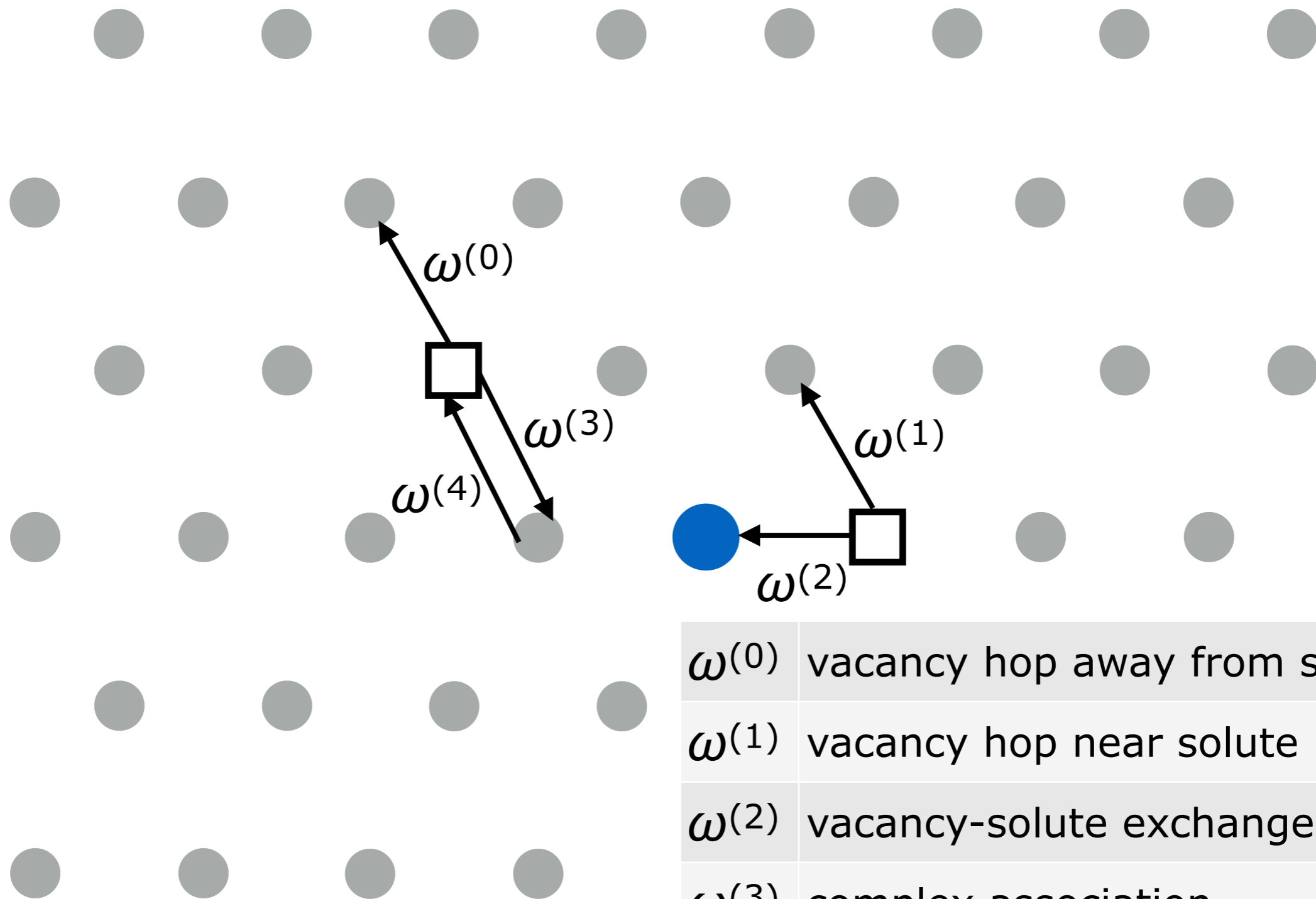
**Example:** vacancy-mediated diffusion in magnesium

# Uncertainty in DFT input data for Mg

1. Use three distinct exchange correlation treatments:
  - Local-density approximation (LDA): Ceperley-Alder
  - Generalized gradient approximation (GGA): PBE
  - Meta-GGA: SCAN
2. Use PAW-PBE relaxed positions as initial guess
  - Scale supercell dimensions to LDA or SCAN lattice constants
  - Compute energy and forces
  - Approximate relaxation energy with PBE lattice Green function:  
 $\Delta E = -\mathbf{f} \cdot (D^{-1}) \cdot \mathbf{f} / 2$
  - Gives correct energies with < 5 meV error

Energy [eV]	PBE	LDA	SCAN
Vacancy formation energy	0.814	0.852	0.948
Pyramidal activation energy	0.417	0.429	0.489
Basal activation energy	0.397	0.409	0.481

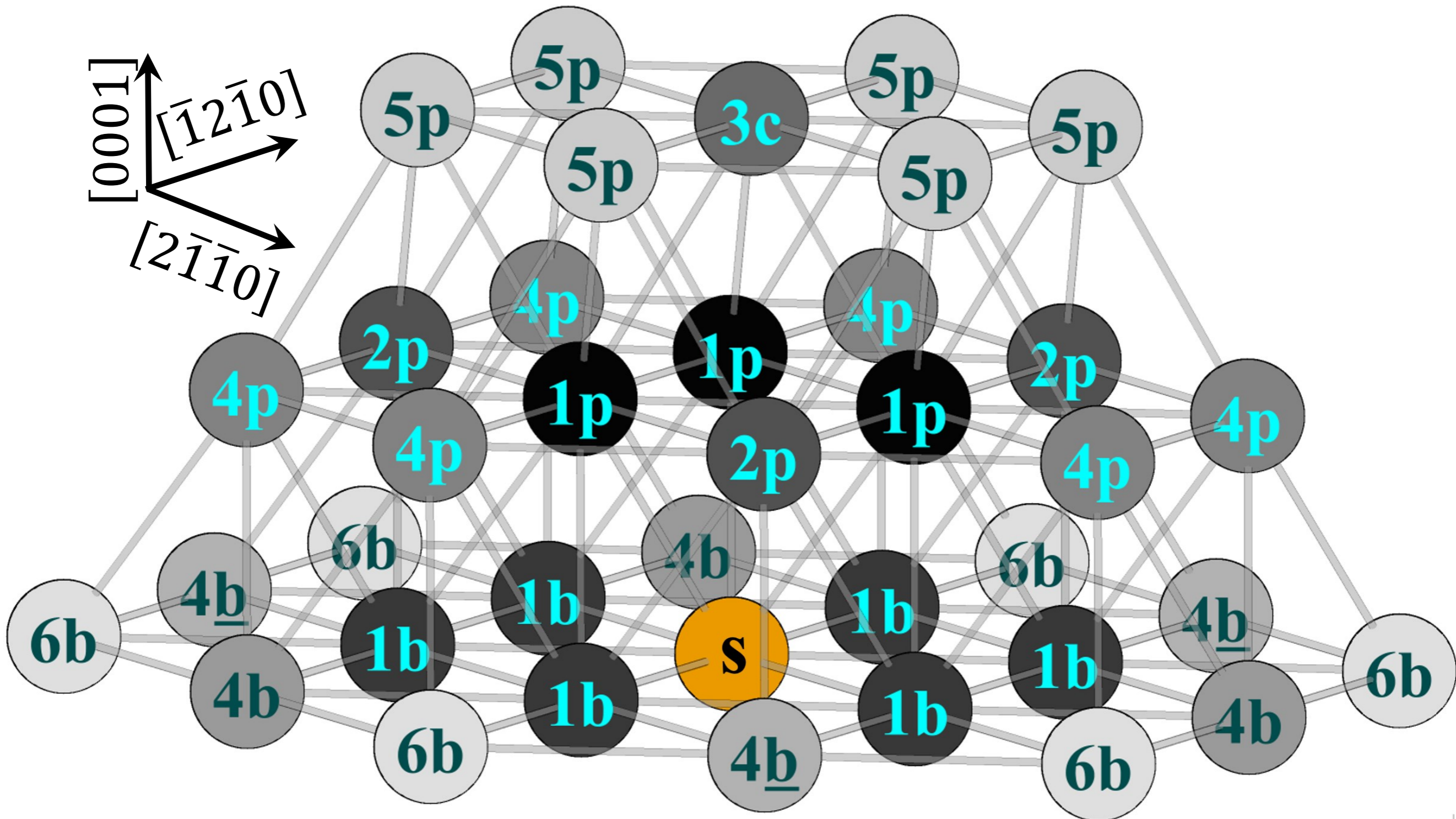
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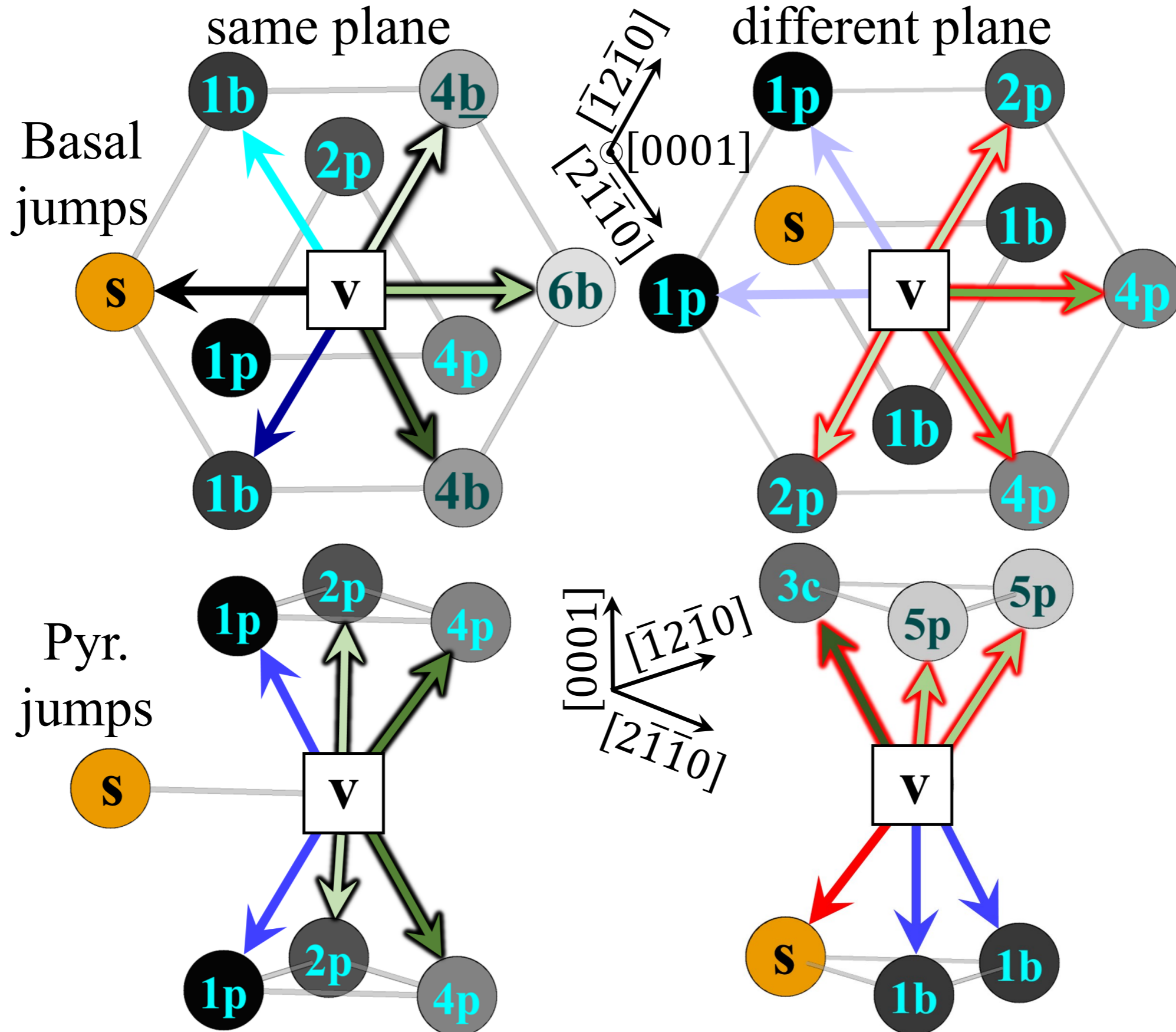
# Vacancy-solute complexes in HCP

- 2 different first-neighbor complexes: 1b 1p
- 7 different "next jump" complexes: 2p 3c 4b 4 $\bar{b}$  4p 5p 6b



# Vacancy jumps near a solute in HCP

Solute and vacancy (1b) Solute and vacancy (1p)

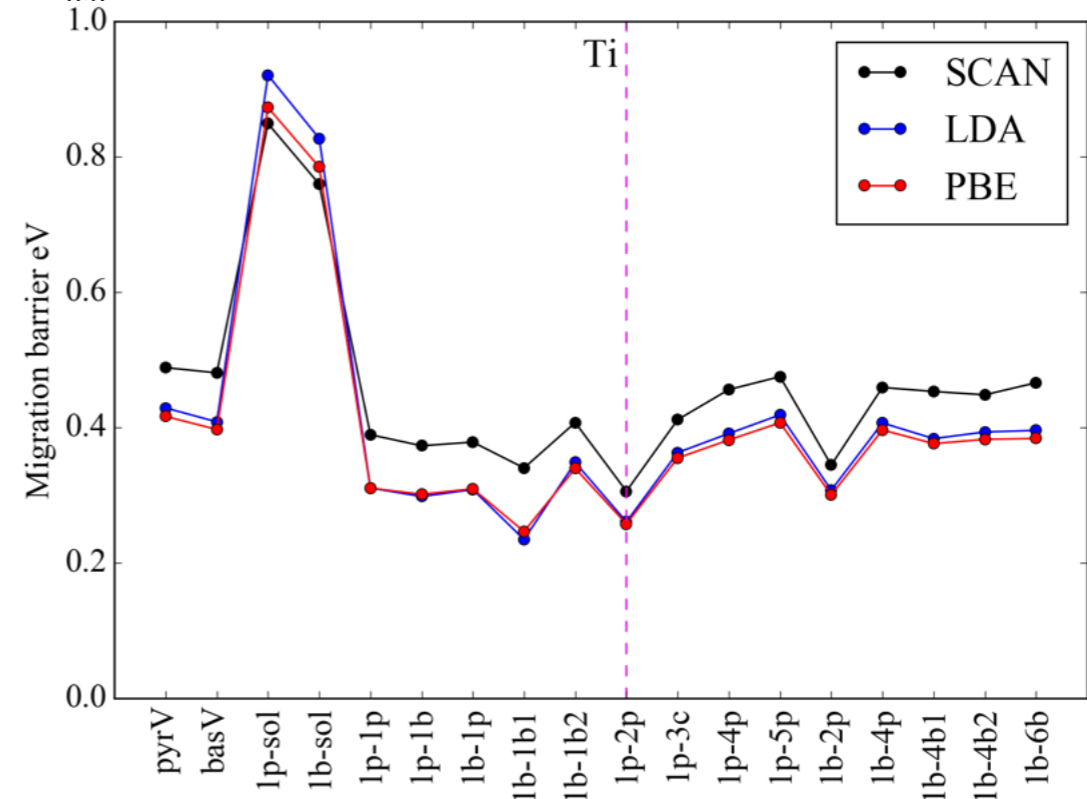
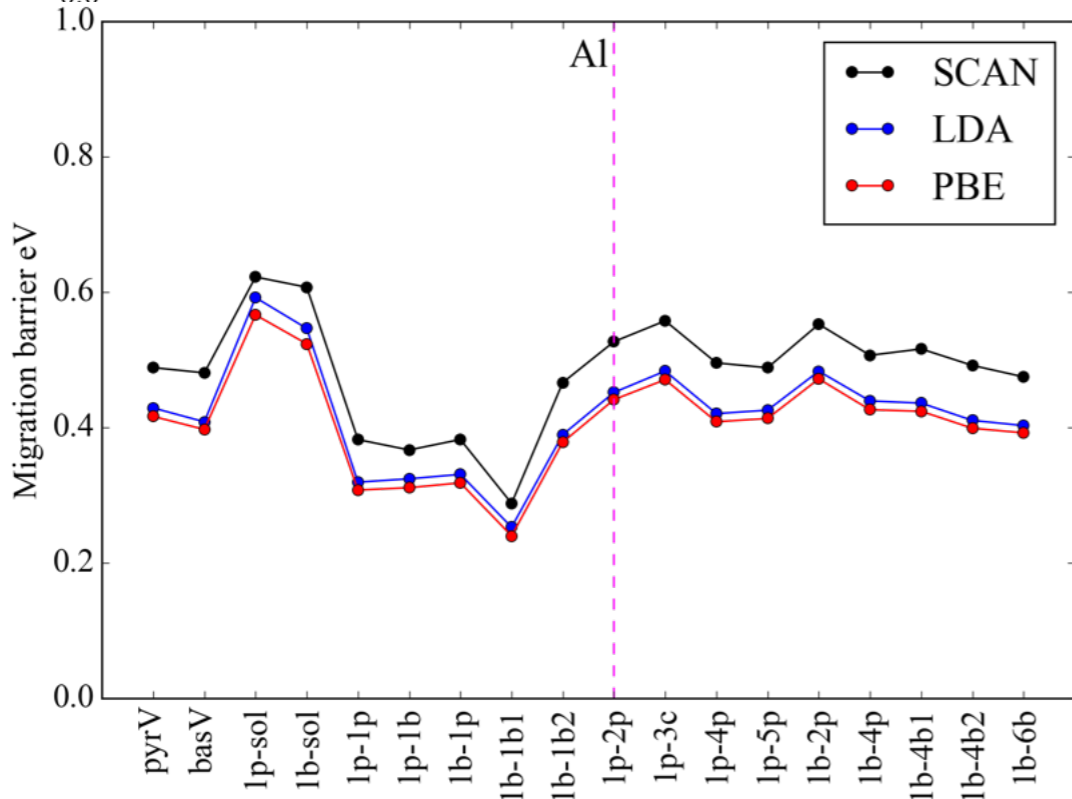
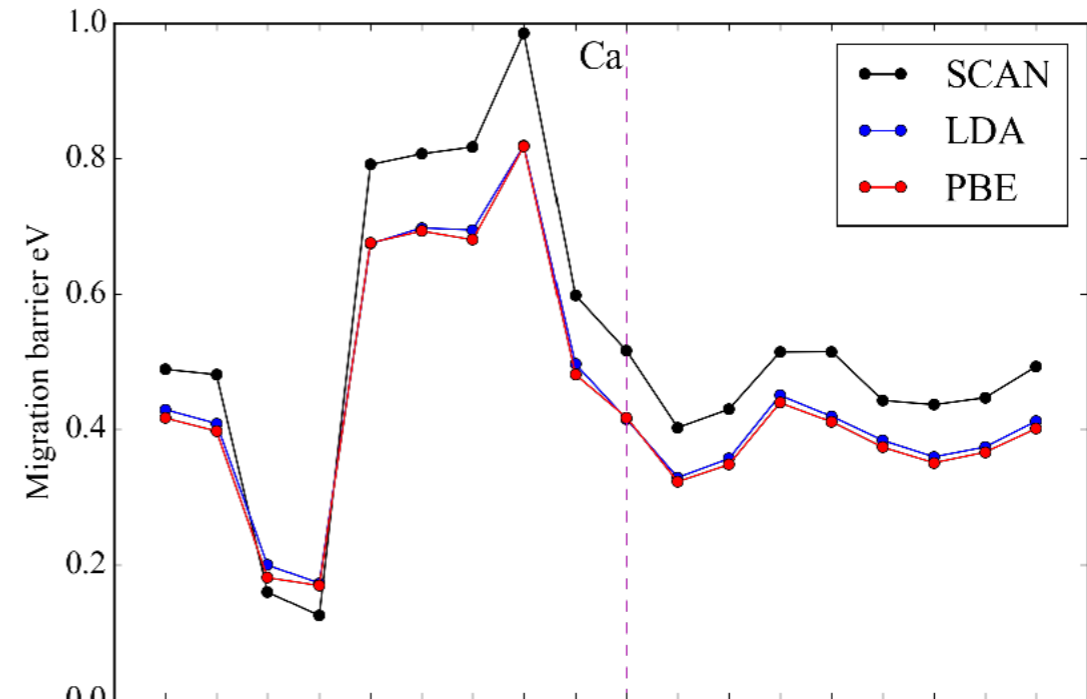
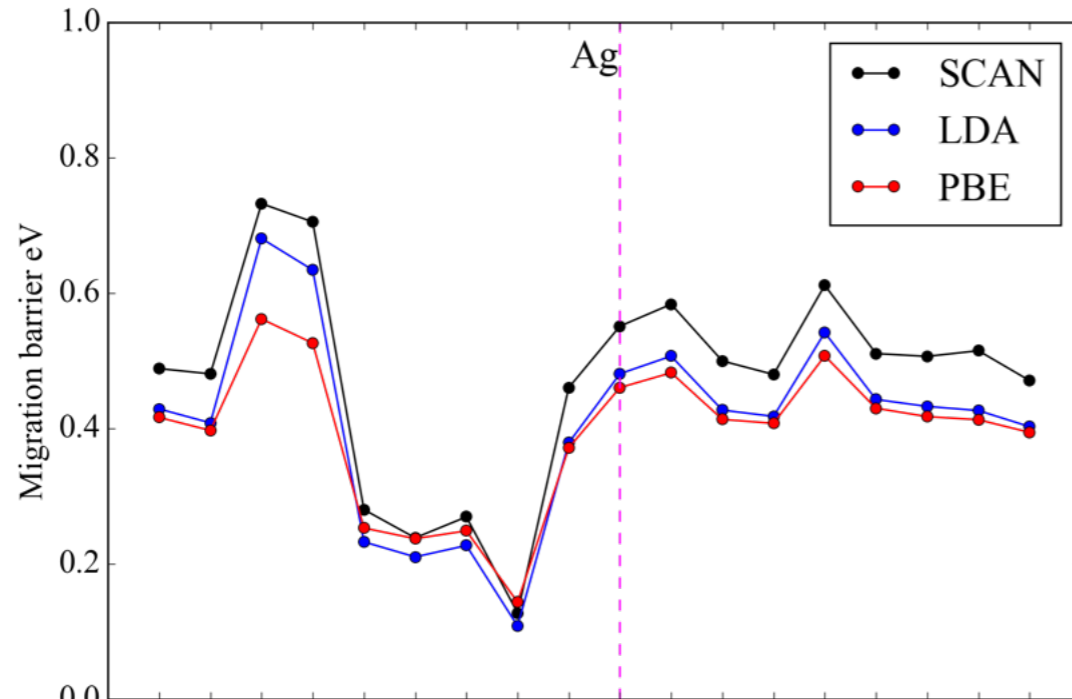




# Uncertainty in DFT input data for Mg

1. Use three distinct exchange correlation treatments:

- Local-density approximation (LDA): Ceperley-Alder
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# Uncertainty in DFT input data for Mg

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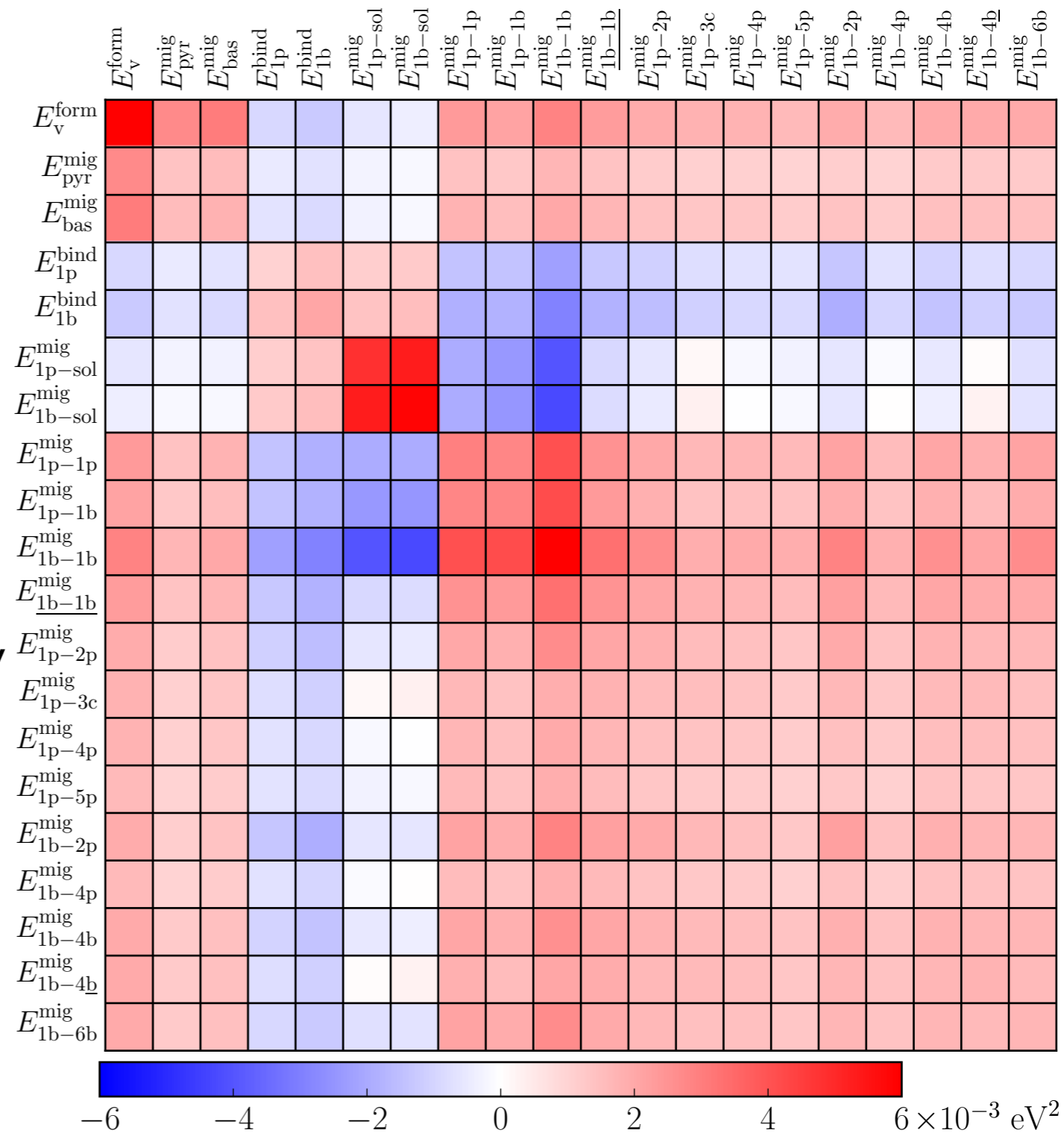
- Local-density approximation (LDA): Ceperley-Alder
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2. Differences due to XC similar for different chemistries:

**treat as random variation**

## Covariance matrix:

- Diagonal: variance around mean
- Off-diagonal: (anti)correlation between quantities across different chemistries
- Source of variability: solute chemistry and exchange-correlation treatment



# Uncertainty in diffusivity

1. Take variability in DFT input parameters and *sample* possible DFT data
  - Bayesian inference approach
  - Assume multivariate normal distribution: empirical covariances ( $\Sigma$ ) and mean values ( $\underline{\theta}$ )

$$P(\theta) = \det(2\pi\Sigma)^{-1/2} \exp\left[-\frac{1}{2}(\theta - \bar{\theta})\Sigma^{-1}(\theta - \bar{\theta})\right]$$

$$\text{mean value of } f: \quad \langle f(\theta; T) \rangle = \int_{\theta} d\theta P(\theta) f(\theta; T)$$

$$\text{probability distribution of } f: \quad P^f(f_0; T) = \int_{\theta} d\theta P(\theta) \delta(f_0 - f(\theta; T))$$

# Uncertainty in diffusivity

1. Take variability in DFT input parameters and *sample* possible DFT data
  - Bayesian inference approach
  - Assume multivariate normal distribution: empirical covariances ( $\Sigma$ ) and mean values ( $\underline{\theta}$ )
2. Distribution of predictions:
  - Diffusivities follow approximately log-normal distributions
  - Drag ratios ( $L^{SV}/L^{SS}$ ) follow more complex distribution
3. Computational details:
  - Use Gaussian-Hermite quadrature for vacancy parameters for computational efficiency
  - Multivariate normal distribution samples for solute-vacancy parameters

$$\theta = \begin{pmatrix} \theta^v \\ \theta^s \end{pmatrix}, \quad \Sigma = \begin{pmatrix} \Sigma^{vv} & \Sigma^{vs} \\ \Sigma^{sv} & \Sigma^{ss} \end{pmatrix} \quad P(\theta) = P^v(\theta^v)P^s(\theta^s|\theta^v)$$

# Uncertainty in diffusivity

## 3. Computational details:

- Use Gaussian-Hermite quadrature for vacancy parameters for computational efficiency
- Multivariate normal distribution samples for solute-vacancy parameters

$$\theta = \begin{pmatrix} \theta^v \\ \theta^s \end{pmatrix}, \quad \Sigma = \begin{pmatrix} \Sigma^{vv} & \Sigma^{vs} \\ \Sigma^{sv} & \Sigma^{ss} \end{pmatrix} \quad P(\theta) = P^v(\theta^v)P^s(\theta^s|\theta^v)$$

- Both  $P^v(\theta^v)$  and  $P^s(\theta^s|\theta^v)$  are multivariate normal distributions.
  - $P^v$  has mean  $\bar{\theta}^v$  and covariance  $\Sigma^{vv}$
  - $P^s$  mean depends on  $\theta^v$  but covariance is independent:

$$\text{mean : } \bar{\theta}^s + \Sigma^{sv}(\Sigma^{vv})^{-1}(\theta^v - \bar{\theta}^v)$$

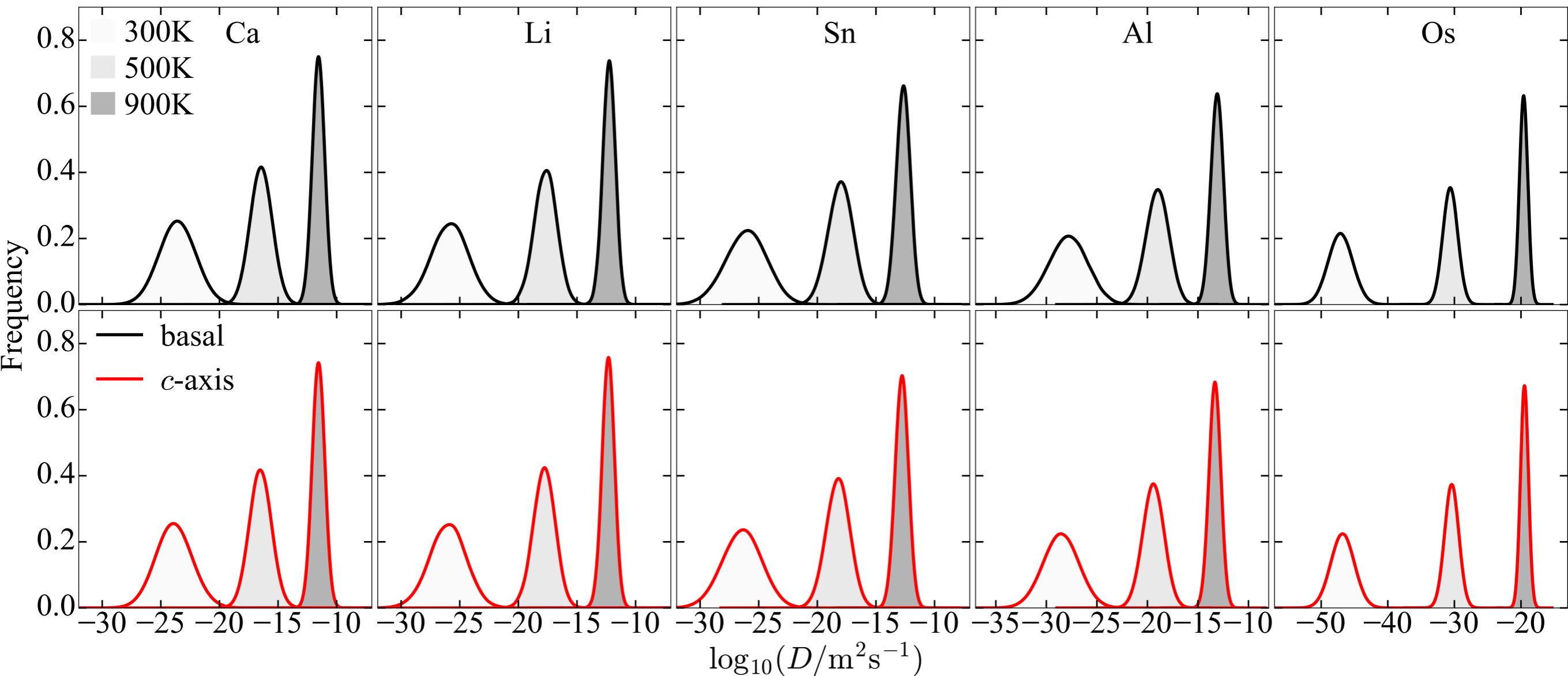
$$\text{covariance : } \Sigma^{ss} - \Sigma^{sv}(\Sigma^{vv})^{-1}\Sigma^{vs}$$

$$\langle f(\theta^v, \theta^s; T) \rangle = \int_{\theta^v} d\theta^v P^v(\theta^v) \left[ \int_{\theta^s} d\theta^s P^s(\theta^s|\theta^v) f(\theta^v, \theta^s; T) \right]$$

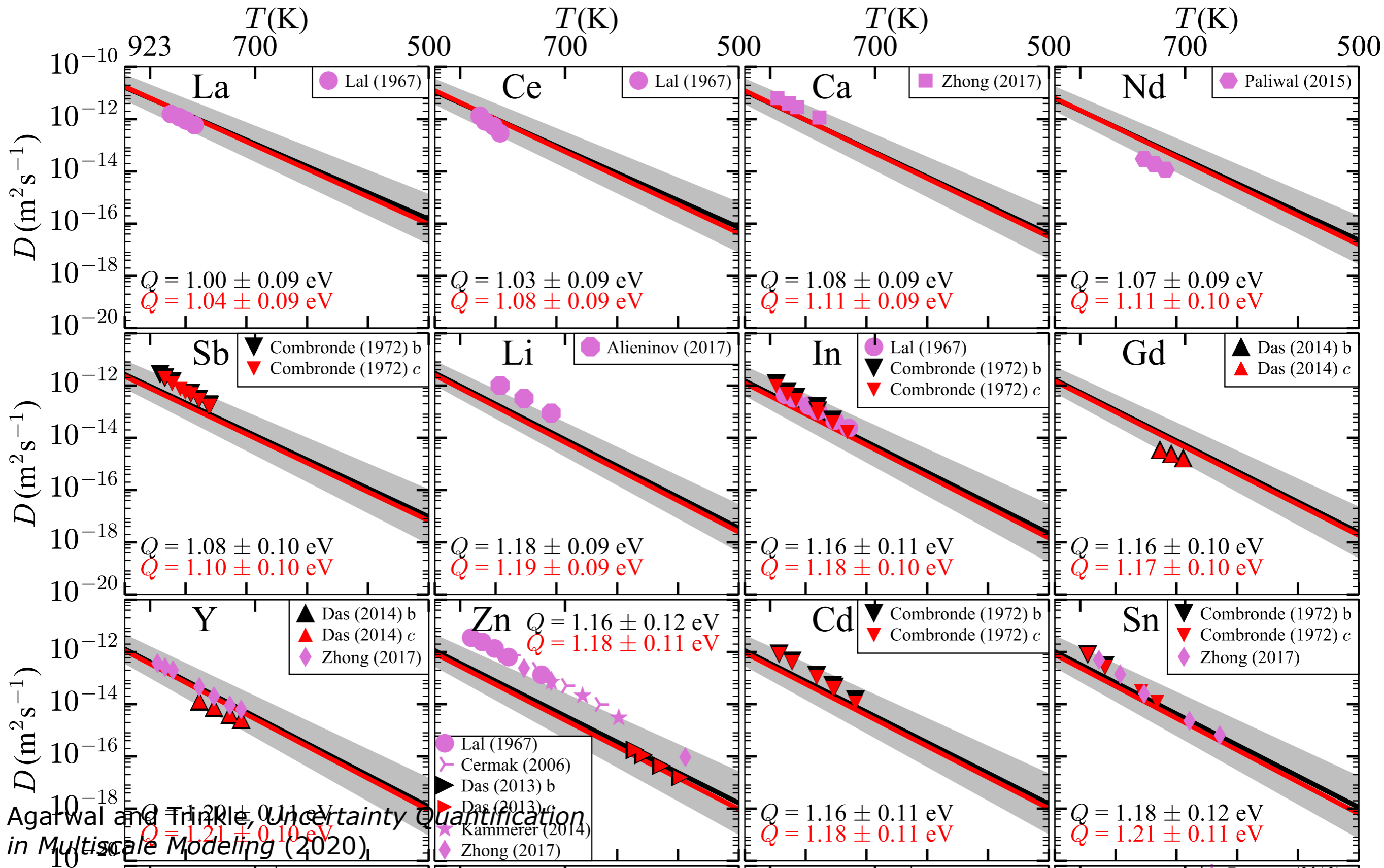
$$\langle f(\theta^v, \theta^s; T) \rangle \approx \sum_{i=1}^{N_{\text{GH}}} \frac{w_i}{N} \sum_{j=1}^N f(\theta^{v,i}, \theta^{s,j}; T)$$

# Uncertainty in diffusivity: histograms of X-Mg

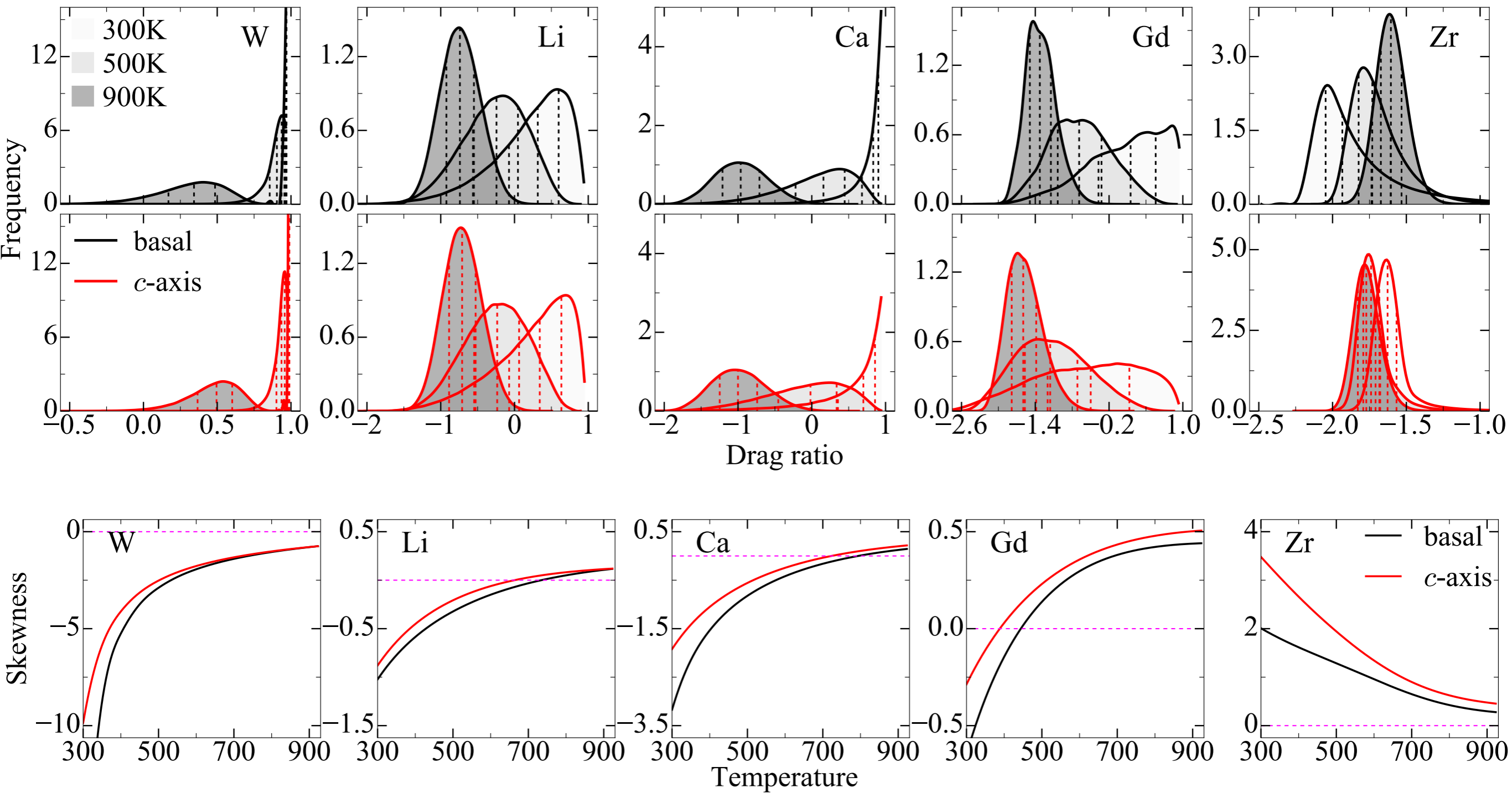
1. Take variability in DFT input parameters and *sample* possible DFT data
  - Bayesian inference approach
  - Assume multivariate normal distribution: empirical covariances and mean values



# Uncertainty in diffusivity: X in Mg vs. experiment

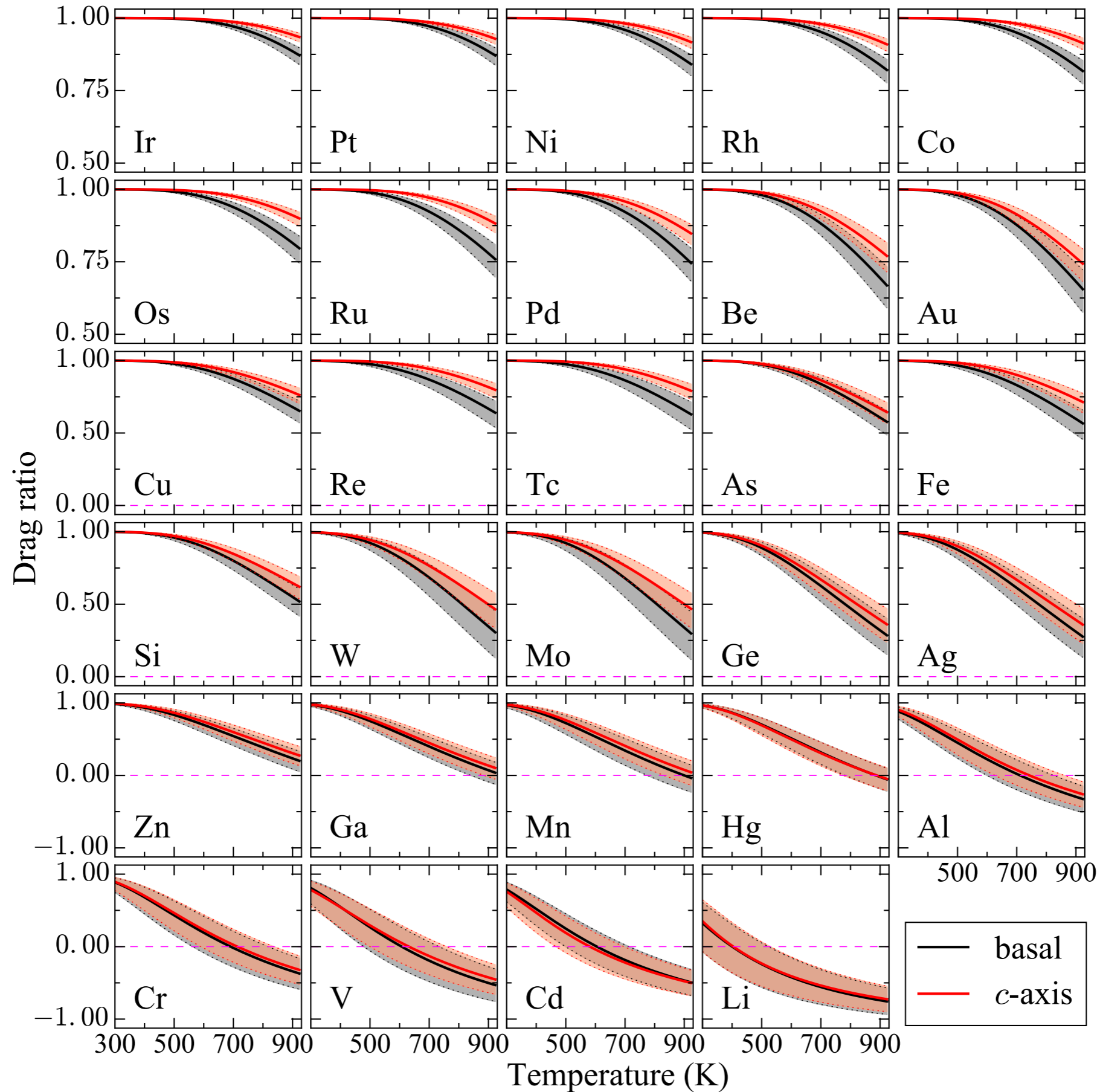


# Uncertainty in drag: histograms of X-Mg

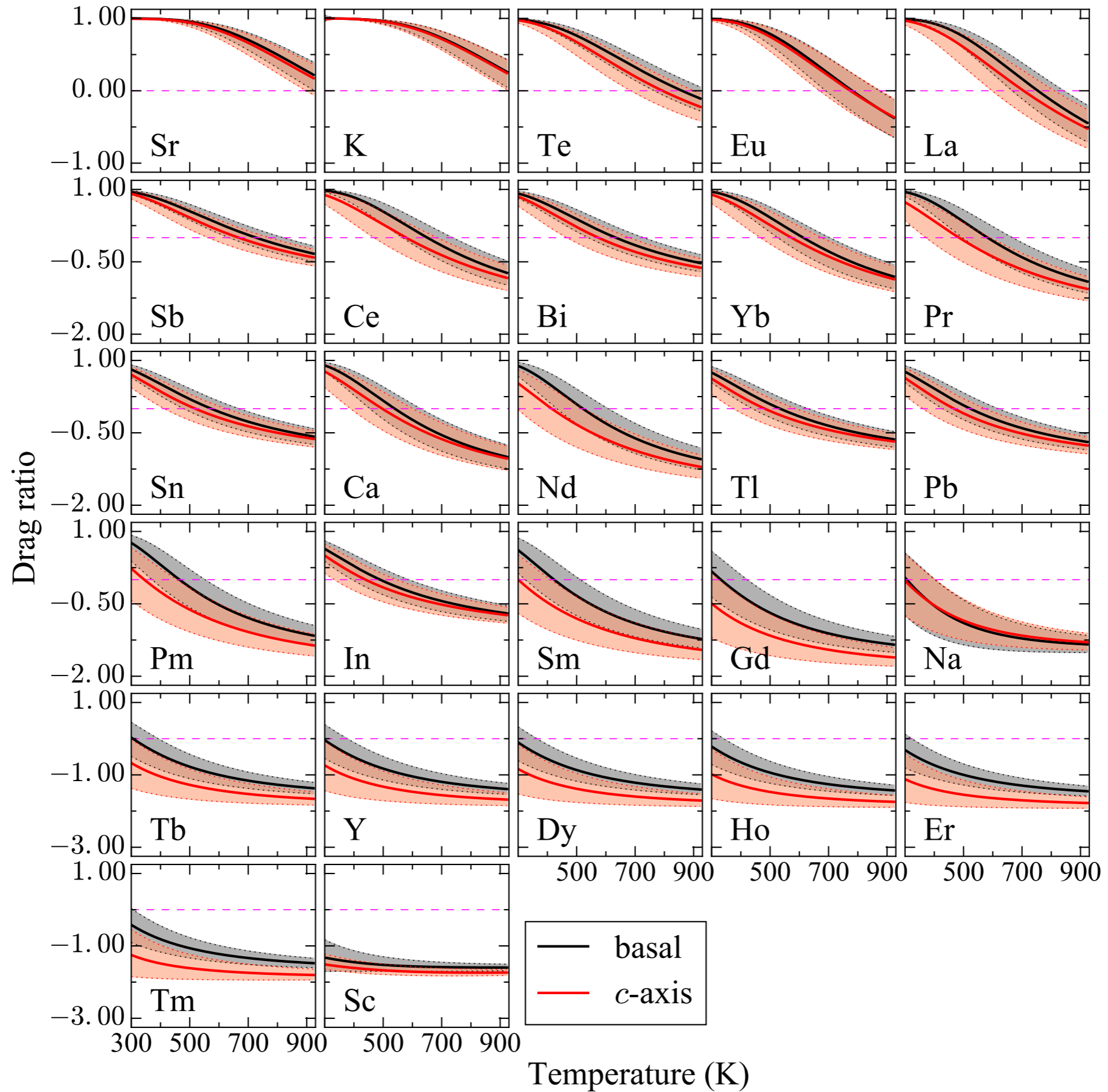




# Uncertainty in drag: X-Mg



# Uncertainty in drag: X-Mg



# Uncertainty quantification for solute transport modeling

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**Motivation:** who do we blame when experiments and theory don't agree?

- **Vacancy-mediated diffusivity**

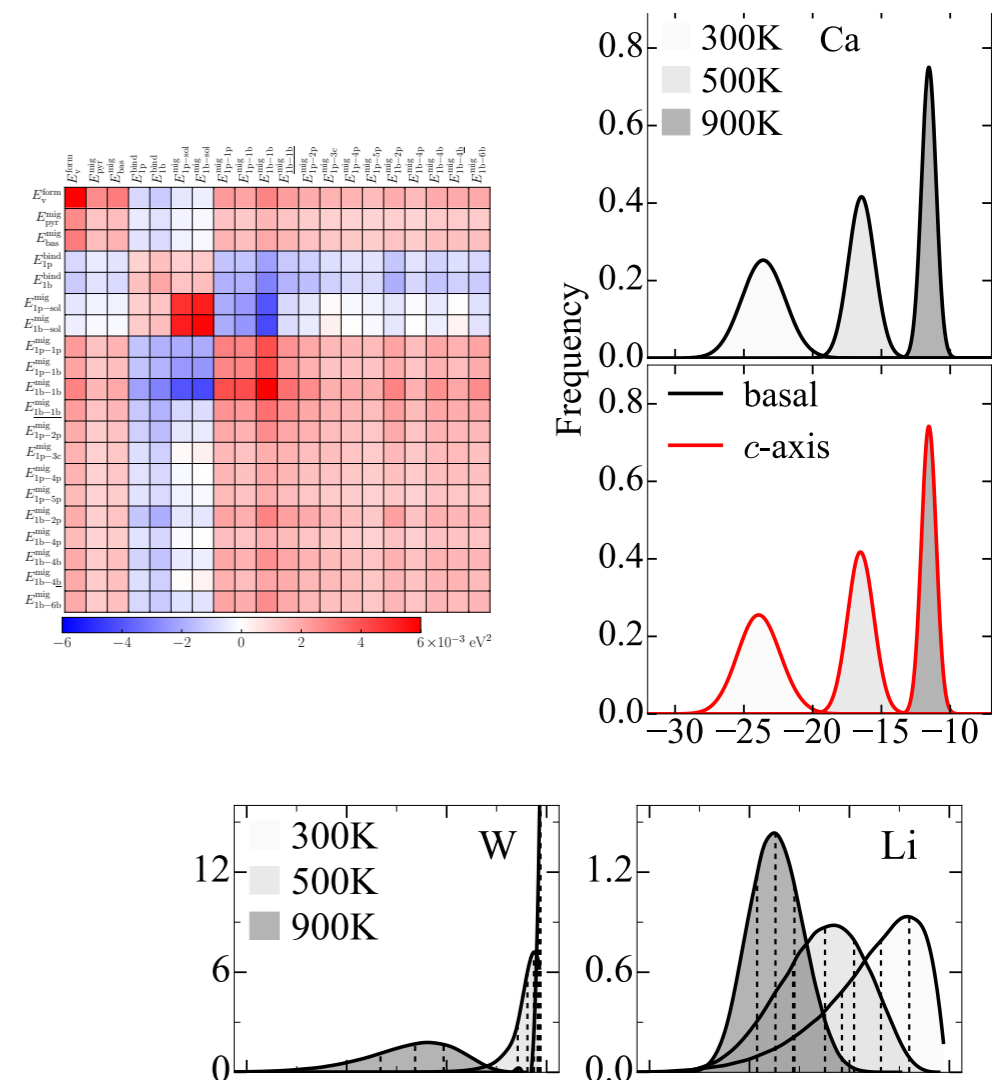
- Quantifying uncertainty in DFT values
- Quantifying uncertainty in diffusivity when there are multiple activated processes at play

- **Applying error bars to DFT predictions**

- Future calculations can include DFT errors
- Derivatives sufficient for diffusivity UQ

- **Possible correlations across solutes**

- Fast substitutional solute diffusivity dominated by vacancy processes: may look like systematic errors
- Slow substitutional solutes and interstitial solute diffusivity dominated by solute processes



Propagating uncertainty from first-principles calculations via analytic models of mass transport can quantify uncertainty in diffusivity predictions.



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